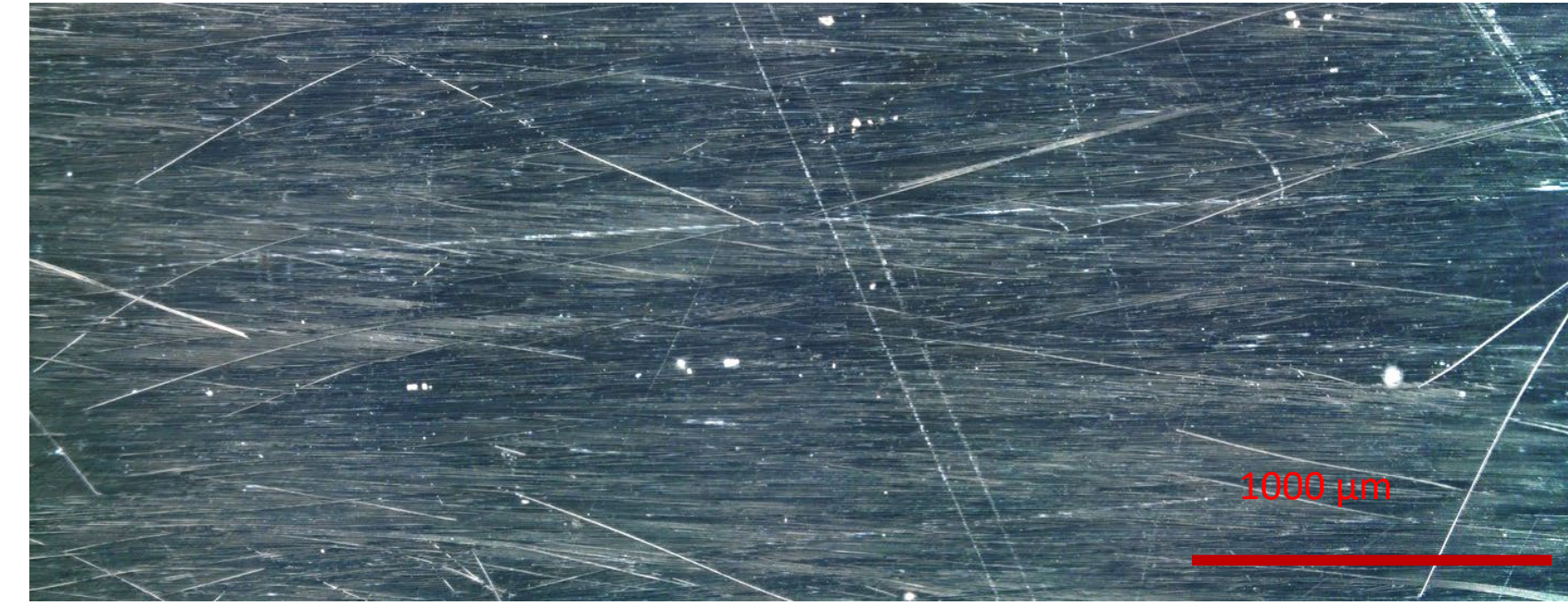
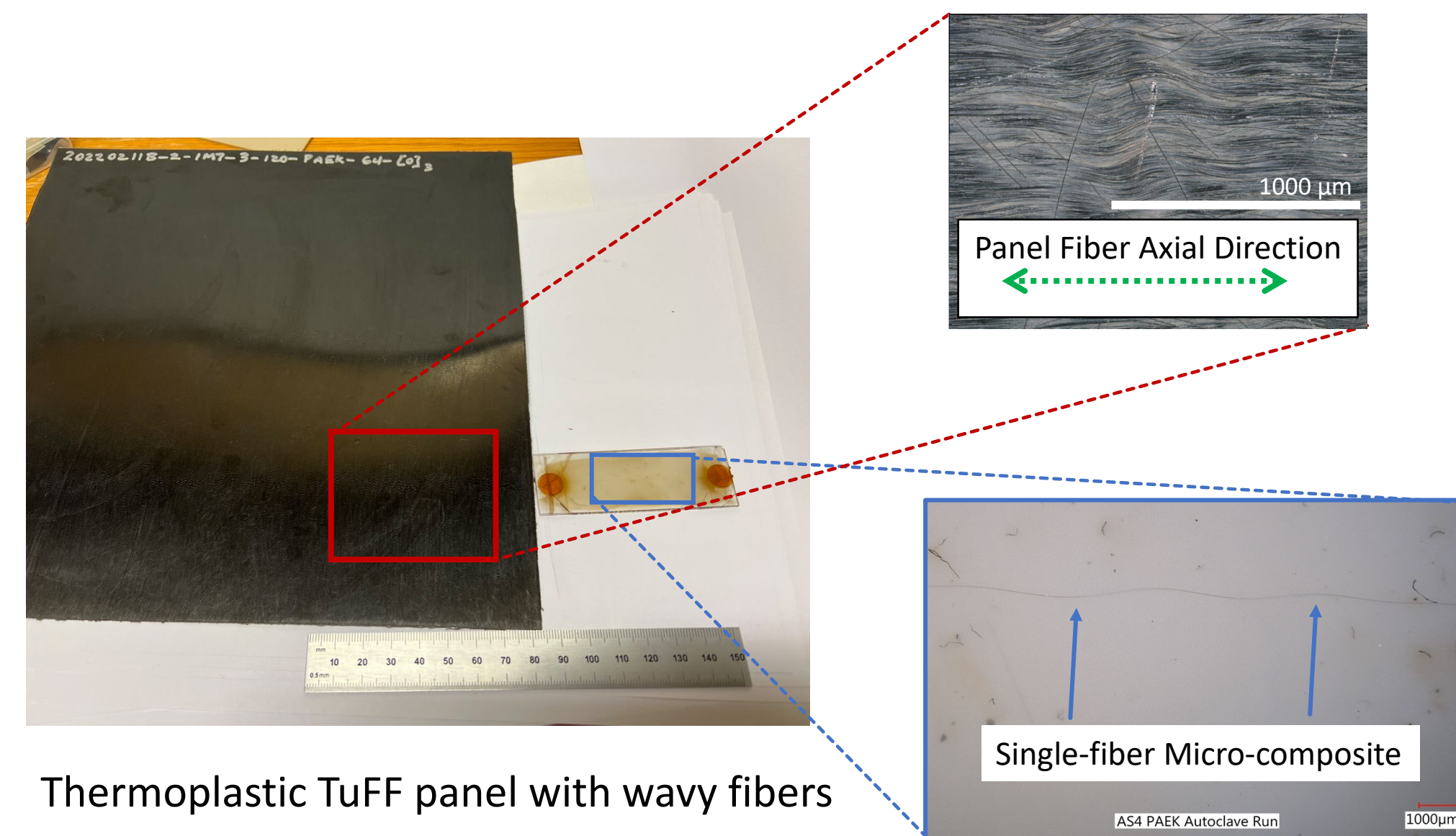


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Introduction



- Poly-ether-imide (PEI) matrix TuFF does not exhibit waviness – fibers are straight



- Semi-crystalline low-melt poly-aryl-ether-ketone (LM-PAEK) matrix TuFF panels exhibit in-plane waviness after consolidation in autoclave processing conditions shown in the figure below
- Also shown is a wavy single fiber micro-composite that was processed alongside the composite panel

Fiber waviness leads to:

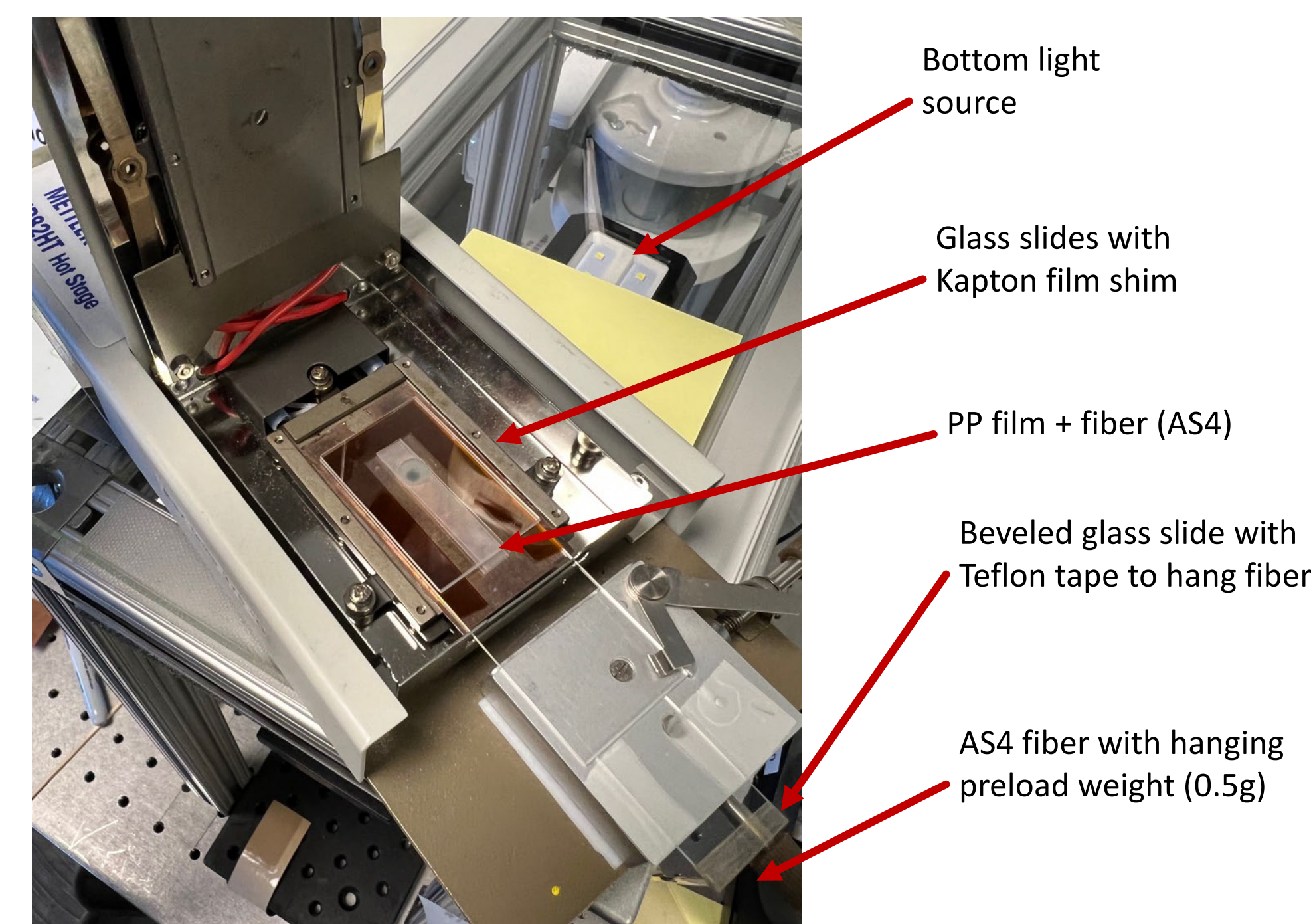
- Increased shear loading of the fiber and matrix interface (IFSS)
- Reduction in modulus and strength
- Larger variability in material static and fatigue performance

Objectives:

- Visualize and isolate fiber waviness formation using single fiber micro-composites
- Determine temperature ranges
- Quantify the fiber waviness severity

Experimental

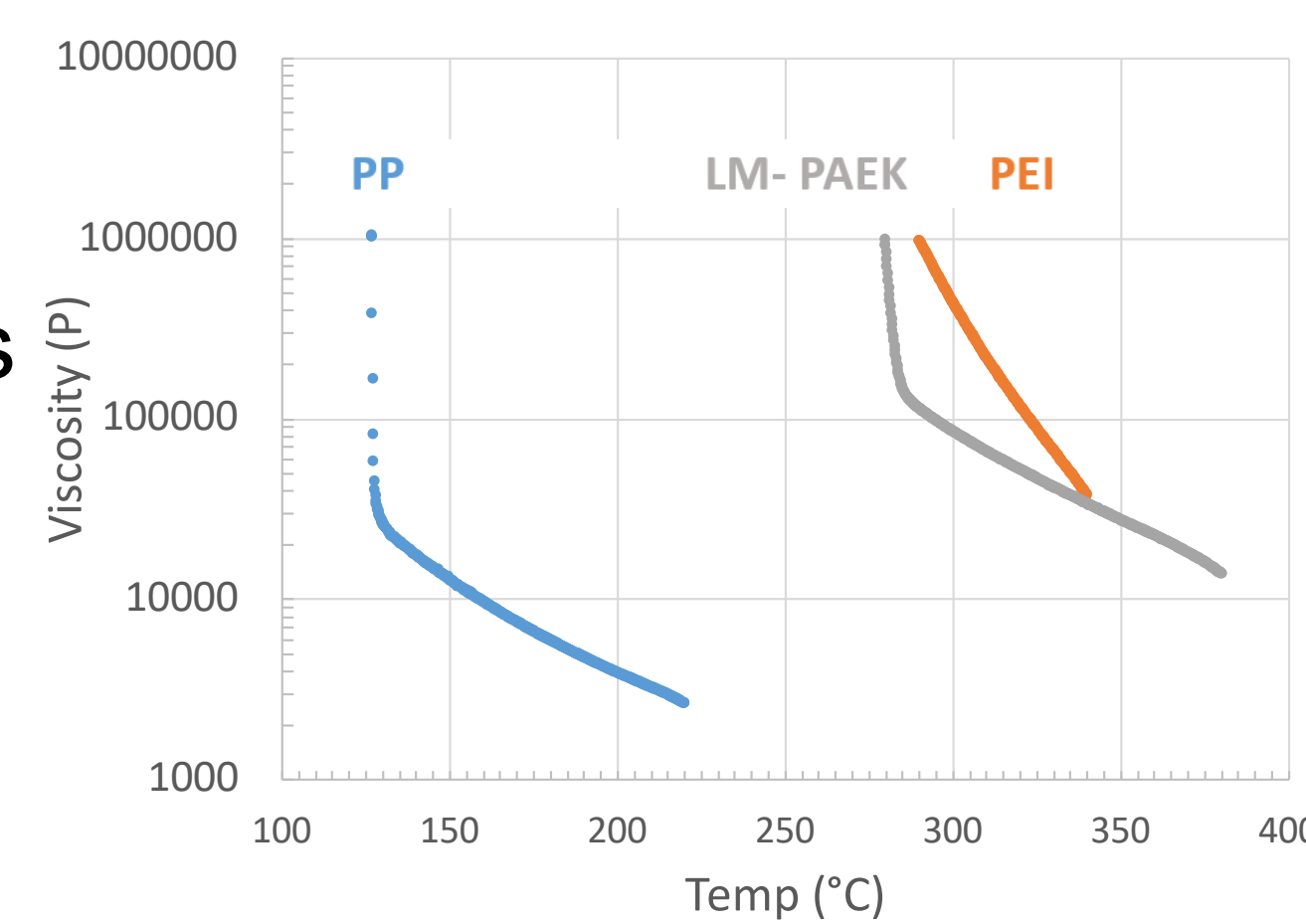
- Utilizing a Mettler hot-stage microscope
 - Controlled cooling rate (20°C/min)
- Iso-tactic polypropylene (PP) $T_{process} = 220\text{ °C}$ used as model matrix material
 - In addition to LM-PAEK and PEI resins
- Single AS4 carbon fibers were separated
- Pre-tension weight (0.5 g) applied during the beginning of the experiments to keep the fibers straight
 - The preload was removed at various temperatures below the process temperature



Mettler Hot-stage Microscope

- Polymer melt viscosity was measured via TA Rheometer

- Viscosity measurements were taken as the polymer cooled from process-melt temperature



Fiber waviness characterization

$$y = a \sin(b * x) \quad \begin{matrix} \lambda = \text{wavelength} \\ a = \text{amplitude} \\ b = \lambda / 2\pi \end{matrix}$$

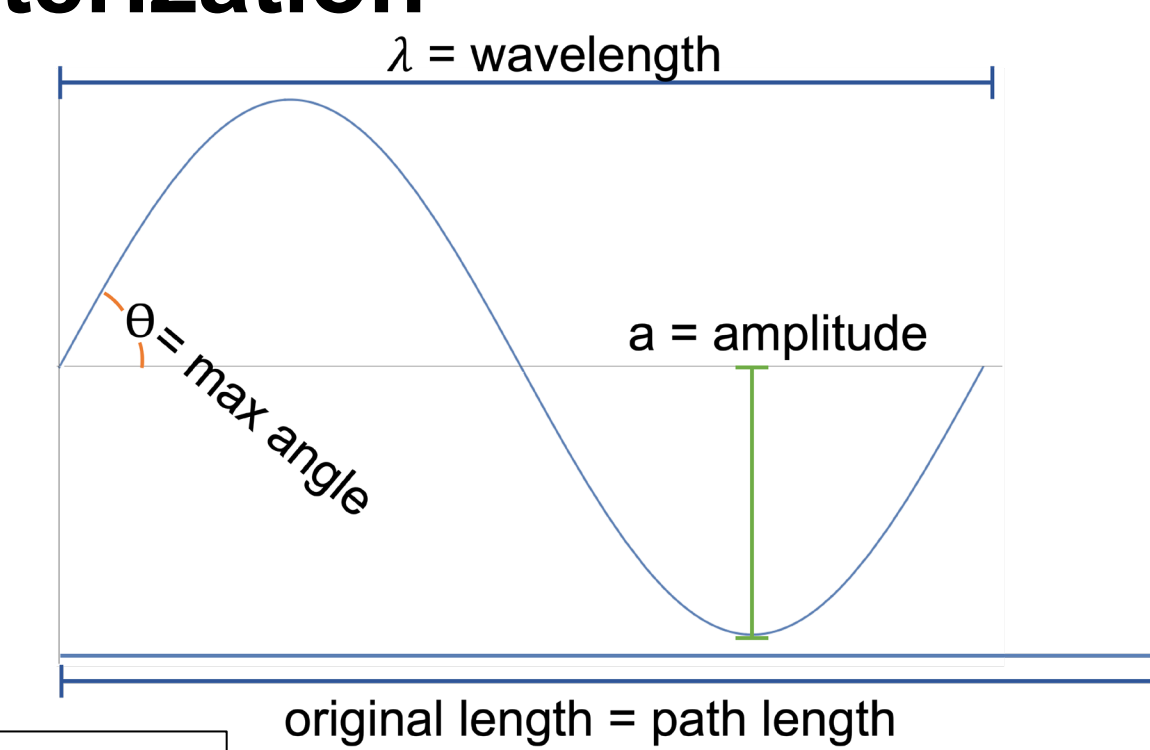
$$\frac{dy}{dx} = a * b \cos(b * x)$$

$$\text{Path length} = \int_0^\lambda \sqrt{1 + \left(\frac{dy}{dx}\right)^2} dx$$

$$\text{Path length} = \int_0^\lambda \sqrt{1 + (a * b \cos(b * x))^2} dx$$

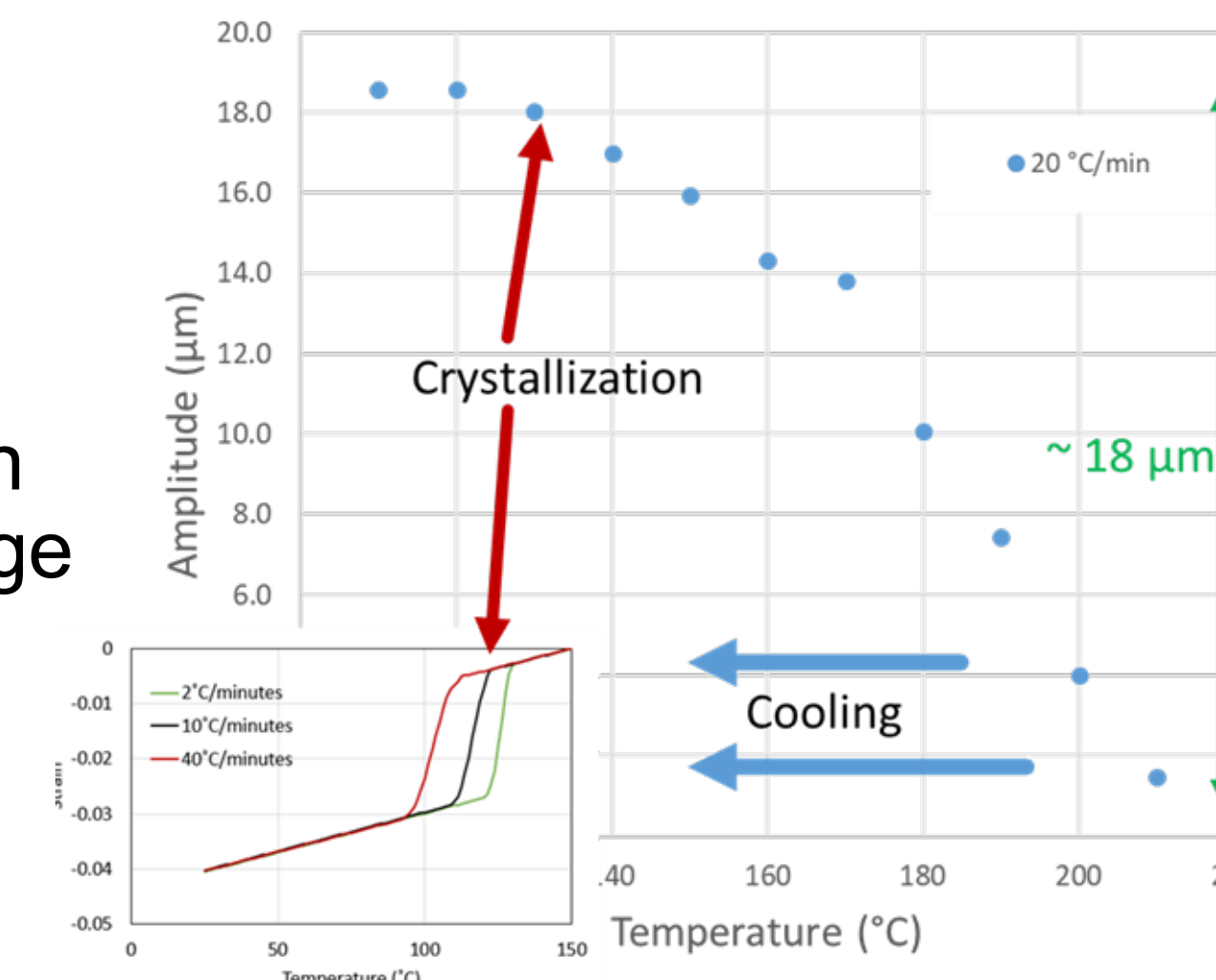
$$\text{Shrinkage strain} = \frac{(\lambda - \text{Path length})}{\text{Path length}}$$

$$\text{Max angle} = \tan^{-1}(a * b \cos(b * x))$$

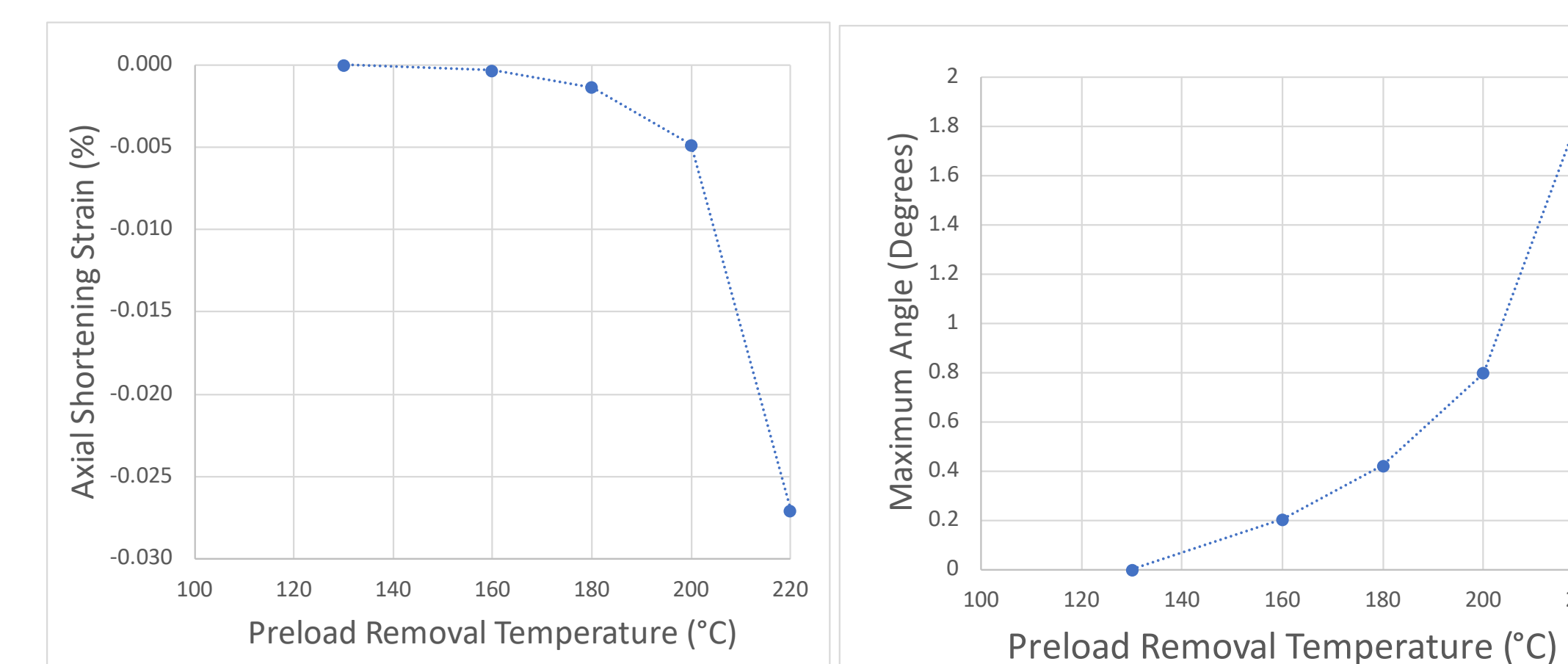
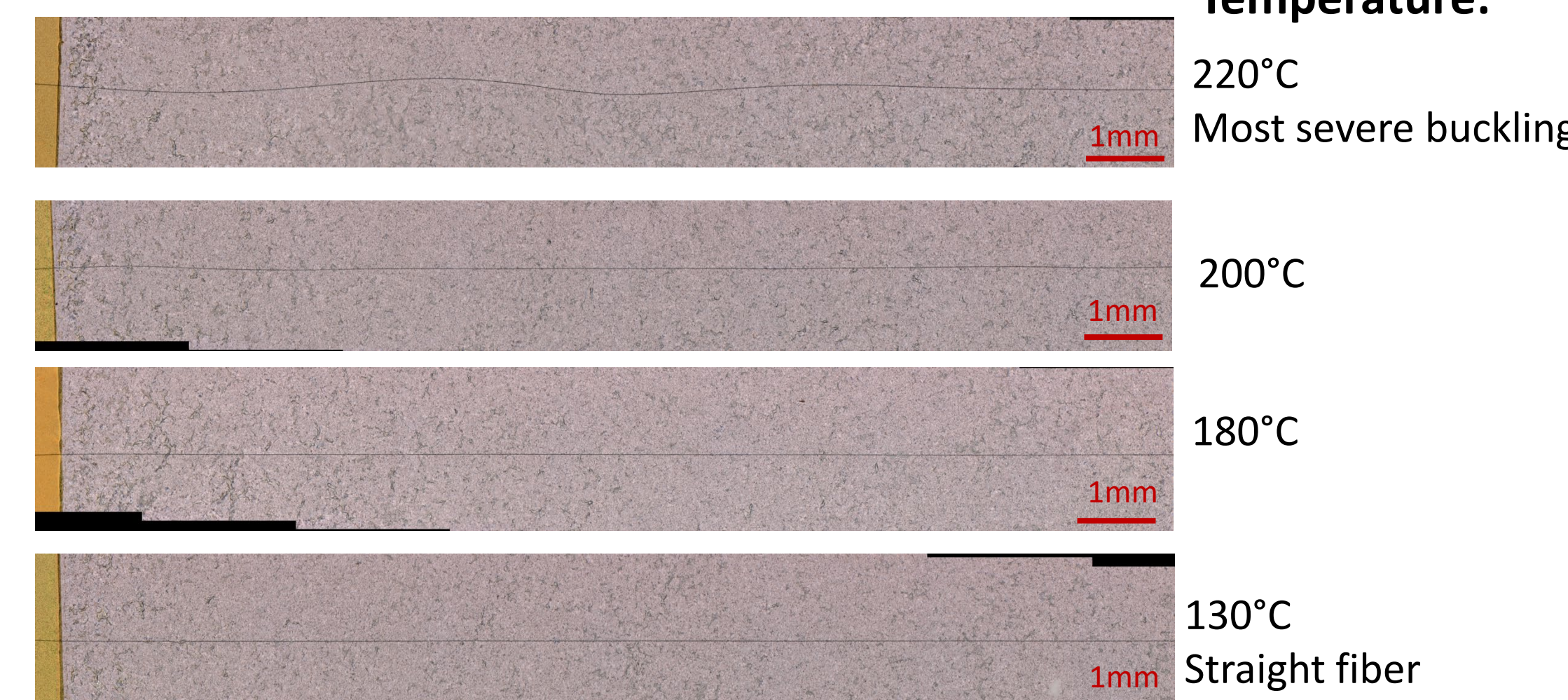


Results

- Observing fiber movement (~18µm) from when the preload is removed at 220 °C through crystallization temperature indicates all **fiber waviness formation occurs during the amorphous melt**
- Fiber movement stops even though the crystallization imparts a large shrinkage strain
- Fiber waviness severity decreases as preload removal temperature is decreased, seen in the micrographs and chart below



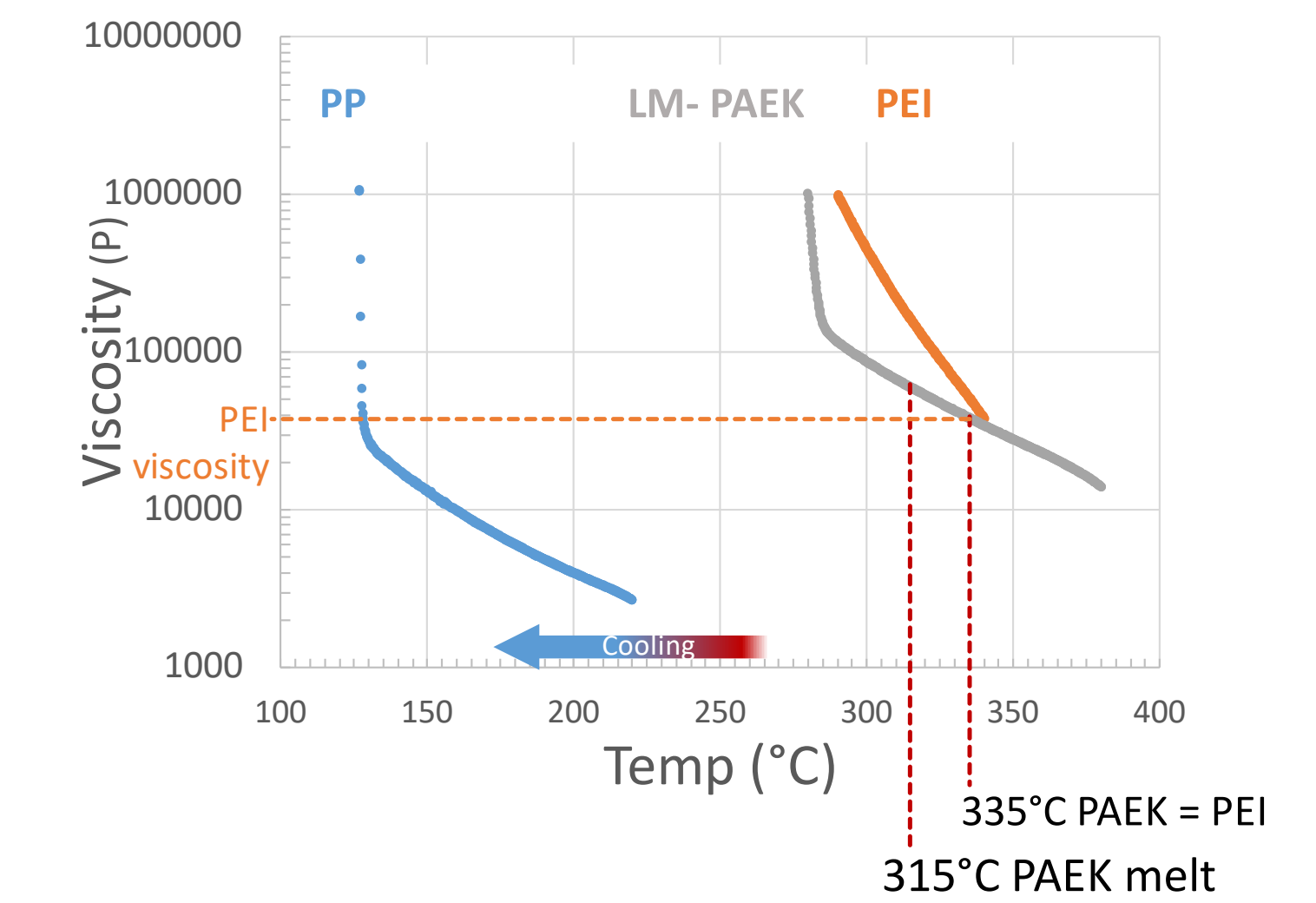
Cooling at 20°C/min



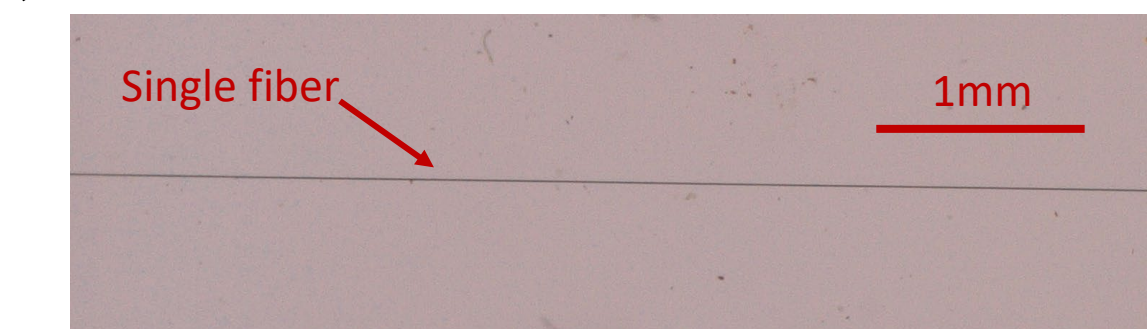
- Shrinkage strain and maximum angle tie together the combinations of wavelength and amplitude the wavy fiber can exhibit
- Amorphous shrinkage and viscosity play a key role in waviness formation
 - Polymer viscosity is low at high $T_{process}$ leading to less resistance to fiber movement
- Critical viscosity of 40,000 P is hypothesized to stabilize the fiber and prevent waviness

Implications on high-performance aerospace matrix materials

- Minimizing process temperature to minimize melt ΔT and maximize viscosity is important to minimizing fiber waviness



Consistency check: **PEI/AS4** micro-composites remain straight, $T=340\text{ °C}$



LM-PAEK:

Removing at 315°C
 $\Delta T = 315 - 288 = 27\text{ °C}$
~ Melt point
Fiber remains straight

Removing at 335°C
 $\Delta T = 335 - 288 = 47\text{ °C}$
Viscosity PEI=LM PAEK
Fiber remains straight

Removing at 380°C
 $\Delta T = 380 - 288 = 92\text{ °C}$
 $T_{process\ max}$
Wavy fiber!

Summary and conclusions

- Novel in situ observation of single fiber waviness formation during the melt has been conducted
- Viscosity and amorphous resin shrinkage are two mechanisms that factor into fiber waviness development
- Observations using a model matrix (PP) have been applied towards LM-PAEK and PEI polymers to control fiber waviness induced by process temperature

Acknowledgements

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