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## Introduction

- Rheology of resin filled aligned discontinuous fiber materials exhibit complex behavior. In order to decouple the effects of strain rate from the polymer and the composite material, a polymer rheology model is required.
- Viscosity of thermosetting resins change with temperature, strain rate, and cure.
- Strain rate effects are rarely accounted for in composite processing since resin typically flows at low viscosity.
- Composite forming is performed at temperatures closer to the glass transition temperature where the resin is viscoelastic.
- This work develops a cure-viscosity model for staged epoxy resins which accounts for the increase in glass transition temperature.

## Methods

### Cure Kinetics Measurements

- Rate of reaction and degree of cure were measured using Differential Scanning Calorimetry (DSC) on a Netzsch DSC 214.
- Temperature was ramped at constant rates.

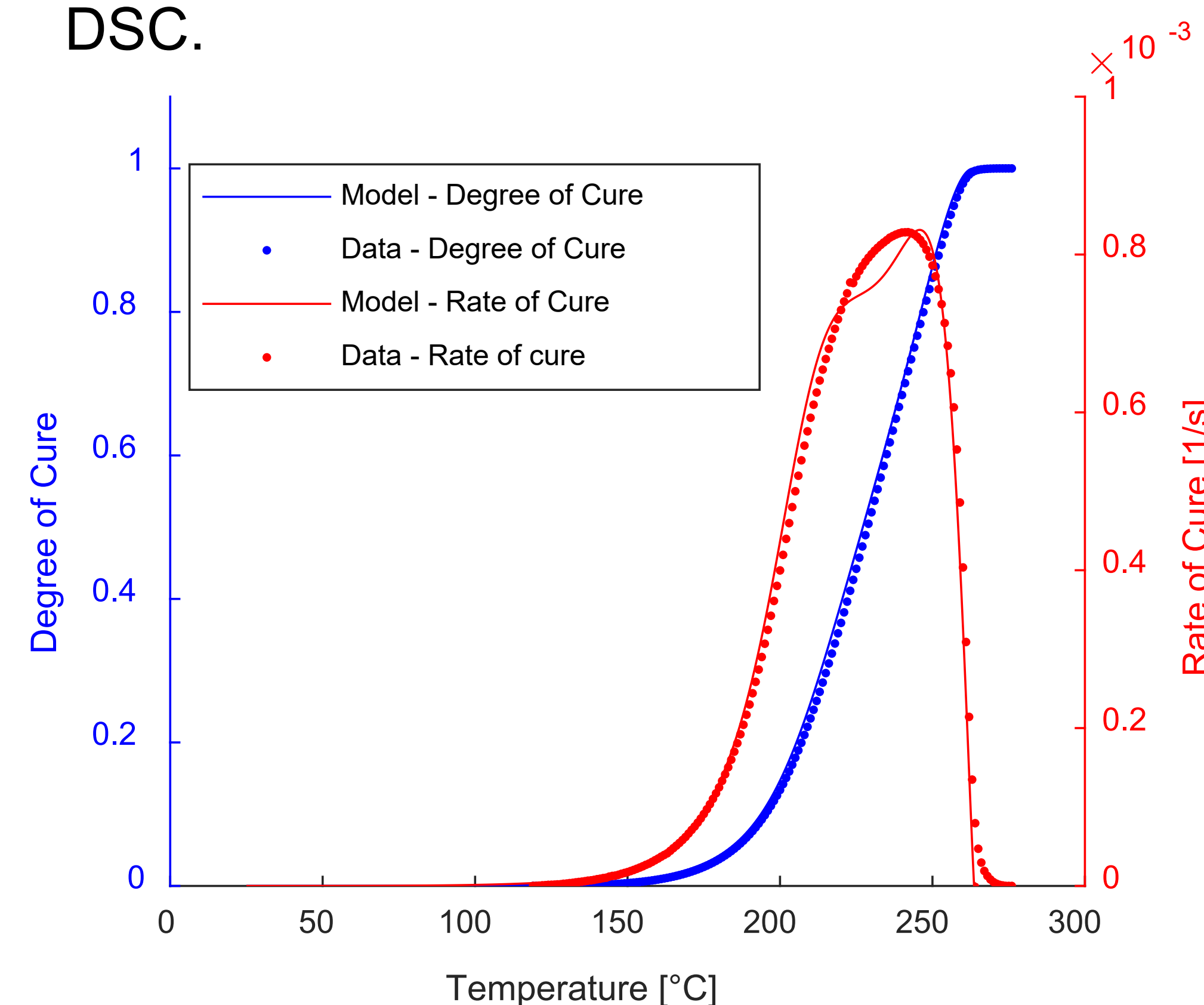
### Viscoelastic Property Measurements

- Viscosity was measured with a TA Instruments Discovery Hybrid Rheometer using 25mm parallel plates
- Temperature was ramped at constant rates of 1- 4 °C/min until gelation.
- Strain amplitude was set to 0.1 oscillating at 1 Hz. Strain amplitude was adjusted to 0.2 at low viscosity to increase resolution.



## Results

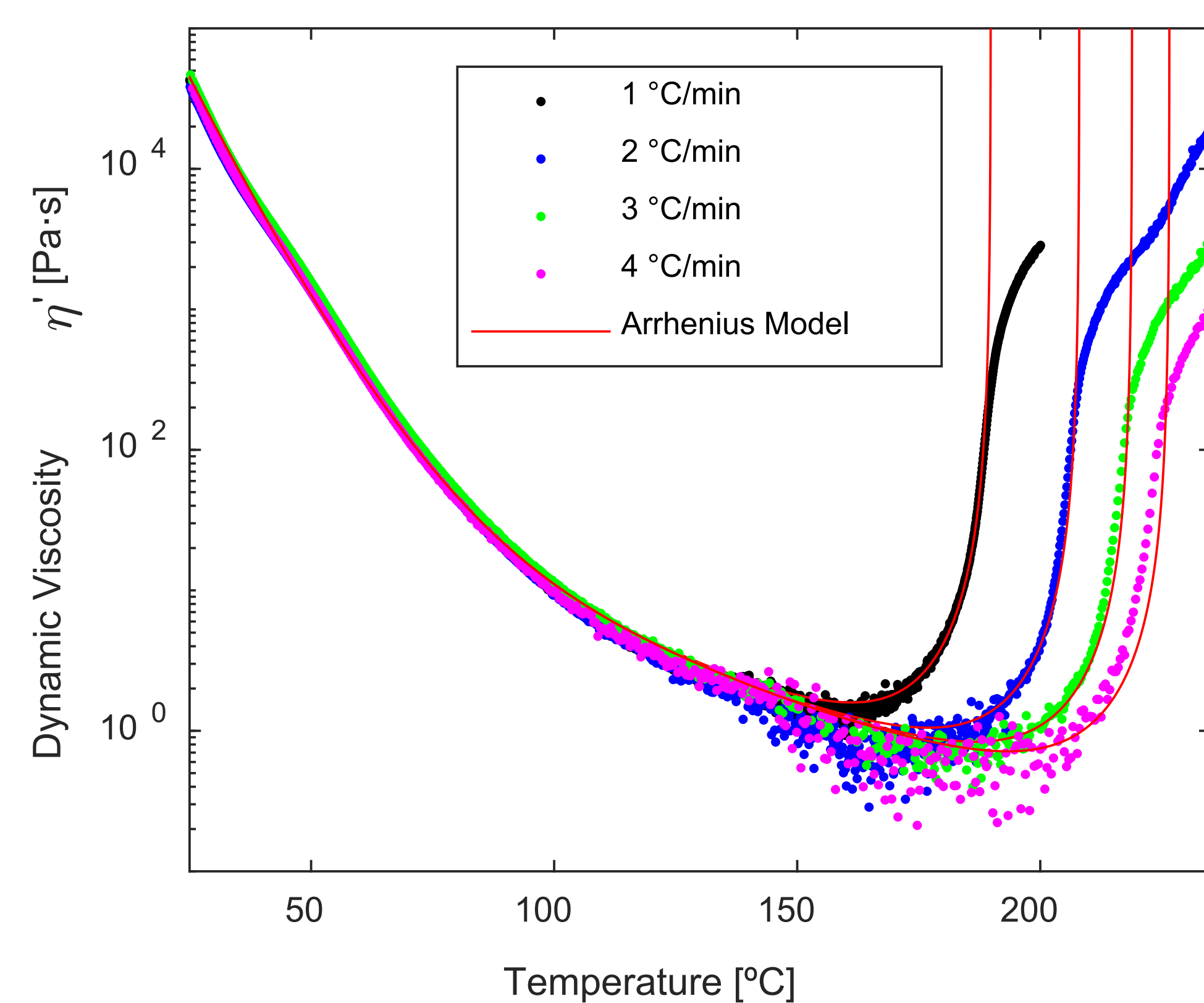
- A cure kinetics model was fit to the rate of reaction and degree of cure data from the DSC.



- The resin cure kinetics model is required to describe the increase in viscosity as degree of cure,  $\alpha$ , approaches gelation.
- The standard Arrhenius viscosity model was constructed to predict the increase in viscosity near the gelation point  $\alpha_g$ .

$$\eta = \eta_1 e^{\frac{E_1}{RT}} + \eta_2 e^{\frac{E_2}{RT}} \left( \frac{\alpha_g}{\alpha_g - \alpha} \right)^A$$

- The Arrhenius model was fit to viscosity data from the rheometer.



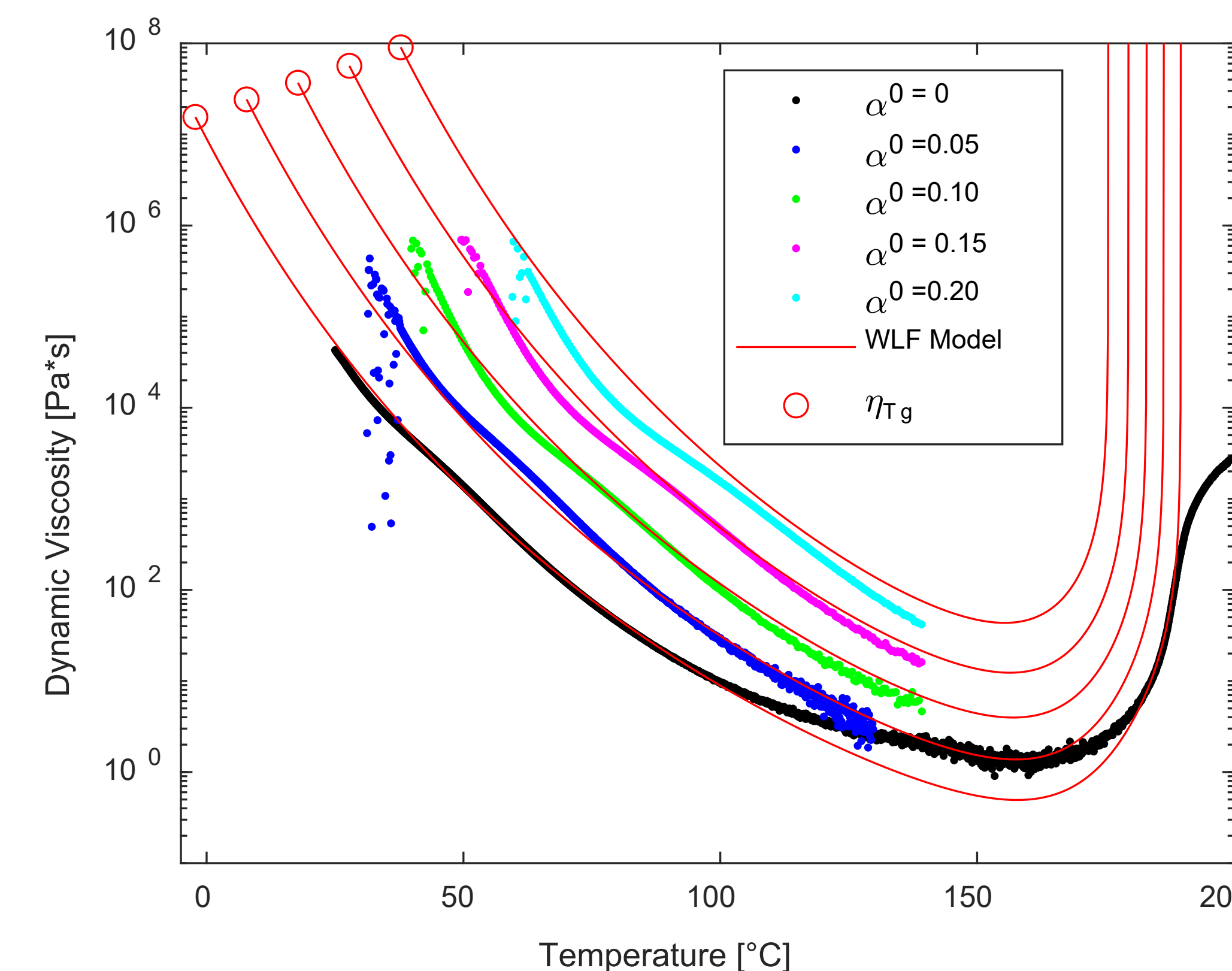
- The Arrhenius model does not work when the resin is staged to a non-zero initial cure.
- To account for the initial cure, the cure dependent glass transition temperature  $T_g$  is used as a reference.

$$T_g = 200\alpha - 3 \text{ [}^\circ\text{C]}$$

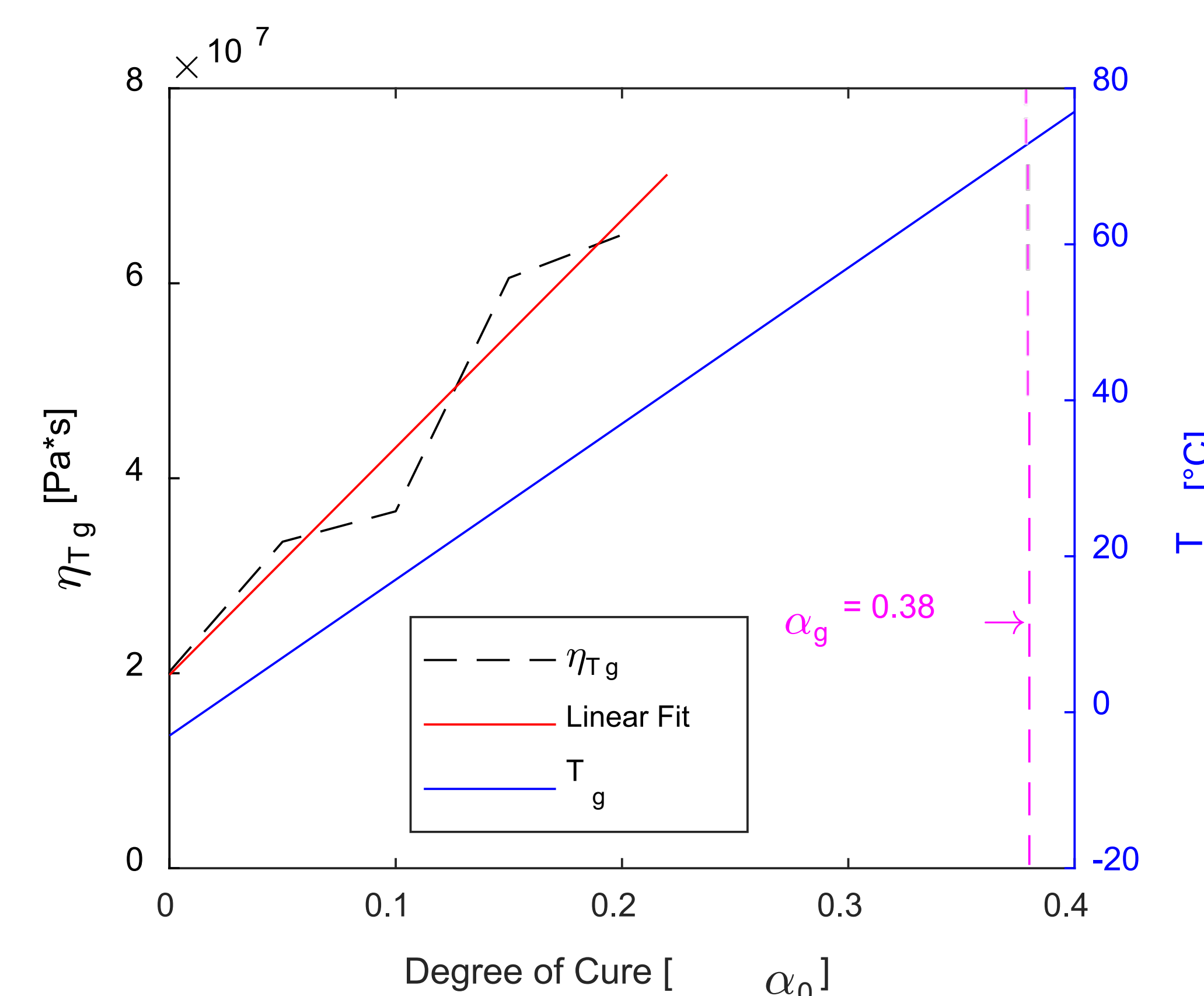
- The William-Landel-Ferry (WLF) viscosity model accounts for changing  $T_g$  making it more accurate at higher initial cures.

$$\eta(T) = \eta_{Tg} \exp\left(\frac{C_1(T - T_g(\alpha))}{C_2 + T - T_g(\alpha)}\right) \left(\frac{\alpha_g}{\alpha_g - \alpha}\right)^A$$

- To study the effect of initial cure, samples were staged to  $\alpha_0 = 5, 10, 15, 20\%$  cure in the rheometer. Viscosity was measured at a 2 °C/min constant temperature ramp.
- In the WLF model  $C_1$  and  $C_2$  were fixed,  $T_g(\alpha_0)$  was used as a reference, and  $\eta_{Tg}$ , the viscosity at the glass transition temperature, was a free parameter.
- Here  $C_1 = 12.9$  and  $C_2 = 101.7$



- $\eta_{Tg}$  and  $T_g(\alpha_0)$  were plotted against degree of cure.  $\eta_{Tg}$  varies linearly with degree of cure like  $T_g(\alpha_0)$ .



## Conclusion and Path Forward

- The Arrhenius viscosity model only applies to the viscosity of B-staged resins.
- The WLF model accounts for the cure dependent glass transition temperature of the resin and thus more accurately models viscosity at low temperatures.
- Future work will investigate the effect of strain rate and employ the WLF model for time shift factors in the viscoelastic regime.

## Acknowledgements

- Thanks to Elizabeth Rock for conducting DSC experimentation.
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