*MAT 162 *MAT COMPOSITE MSC DMG

A PROGRESSIVE COMPOSITE DAMAGE MODEL FOR UNIDIRECTIONAL AND WOVEN FABRIC **COMPOSITES**

UD-CCM Updates on MAT162 USER MANUAL Version 22A-2022

Materials Sciences Corporation (MSC) University of Delaware Center for Composite Materials (UD-CCM)

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*MAT_COMPOSITE_MSC_{OPTION}

Available options include:

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DMG

These are Material Types 161 and 162. These material types may be used to model the progressive failure in composite materials consisting of unidirectional and woven fabric layers subjected to high strain-rate and high pressure loading conditions. The progressive layers failure criteria have been established by adopting the methodology developed by Hashin [1980] with a generalization to include the effect of highly constrained pressure on composite failure. These failure models can be used to effectively simulate fiber failure, matrix damage, and delamination behavior under all conditions – opening, closure, and sliding of failure surfaces. The model with DMG option (material 162) is a generalization of the basic layer failure model of Material 161 by adopting the damage mechanics approach [Matzenmiller et al., 1995] for characterizing the softening behavior after damage initiation. These models require an additional license from Materials Sciences Corporation, which developed and supports these models in collaboration with University of Delaware Center for Composite Materials (UD-CCM).

Card 1	1	2	3	4	5	6	7	8
Variable	MID	RO	EA	EB	EC	PRBA	PRCA	PRCB
Type	A8	F	F	F	F	F	F	F
Card 2	1	2	3	4	5	6	7	8
Variable	GAB	GBC	GCA	AOPT	MACF			
Type	F	F	F	F	I			
Card 3	1	2	3	4	5	6	7	8
Variable	XP	YP	ZP	A1	A2	A3		
Type	F	F	F	F	F	F		
Card 4	1	2	3	4	5	6	7	8
Variable	V1	V2	V3	D1	D2	D3	BETA	
Type	F	F	F	F	F	F	F	
Card 5	1	2	3	4	5	6	7	8
Variable	SAT	SAC	SBT	SBC	SCT	SFC	SFS	S_AB
Type	F	F	F	F	F	F	F	F

Card 6	1	2	3	4	5	6	7	8
Variable	S_BC	S_CA	SFFC	AMODEL	PHIC	E_LIMT	S_DELM	
Type	F	F	F	F	F	F	F	
Card 7	1	2	3	4	5	6	7	8
Variable	OMGMX	ECRSH	EEXPN	CERATE1	AM1			
Type	F	F	F	F	F			

Define the following card if and only if the option DMG is specified.

Card 8	1	2	3	4	5	6	7	8
Variable	AM2	AM3	AM4	CERATE2	CERATE3	CERATE4		
Type	F	F	F	F	F	F		

VARIABLE	DESCRIPTION
MID	Material identification. A unique number or label not exceeding 8 characters must be specified.
RO	Mass density
EA	E _a , Young's modulus - longitudinal direction♥
EB	E _b , Young's modulus - transverse direction♥
EC	E_c , Young's modulus – through thickness direction ullet
PRBA	v_{ba} , Poisson's ratio ba
PRCA	$v_{\rm ca}$, Poisson's ratio ca
PRCB	$v_{\rm cb}$, Poisson's ratio cb
GAB	G _{ab} , shear modulus ab
GBC	G _{bc} , shear modulus bc
GCA	G _{ca} , shear modulus ca

LS-DYNA® notations a, b, & c have the same meaning as for orthotropic material axes notations 1, 2, & 3.

VARIABLE

DESCRIPTION

, , , , , , , , , , , , , , , , , , , ,	
AOPT	Material axes option, see Figure 2.1. (in KEYWORD Manual) EQ.0.0 : locally orthotropic with material axes determined by element nodes as shown in Figure 2.1. Nodes 1, 2, and 4 of an element are identical to the Nodes used for the definition of a coordinate system by *DEFINE_COORDINATE_NODES.
	EQ.1.0 : locally orthotropic with material axes determined by a point in space and the global location of the element center, this is the a-direction.
	EQ.2.0 : globally orthotropic with material axes determined by vectors defined below, as with *DEFINE_COORDINATE_VECTOR.
	LT.0.0: the absolute value of AOPT is a coordinate system ID number (CID on *DEFINE_COORDINATE_NODES, *DEFINE_COORDINATE_SYSTEM or *DEFINE_COORDINATE_VECTOR). Available in R3 version of 971 and later.
MACF	Material axes change flag: EQ.1: No change, default, EQ.2: switch material axes a & b, EQ.3: switch material axes a & c, EQ.4: switch material axes b & c.
XP YP ZP	Define coordinates of point \mathbf{p} for AOPT = 1.
A1 A2 A3	Define components of vector \mathbf{a} for AOPT = 2.
V1 V2 V3	Define components of vector \mathbf{v} for AOPT = 3.
D1 D2 D3	Define components of vector \mathbf{d} for AOPT = 2.
BETA	Layer in-plane rotational angle in degrees.
SAT	Longitudinal tensile strength, S_{aT}
SAC	Longitudinal compressive strength, S_{aC}
SBT	Transverse tensile strength, S_{bT}
SBC	Transverse compressive strength, S_{bC}
SCT	Through thickness tensile strength, S_{cT}
SFC	Crush strength, S_{FC}
SFS	Fiber mode shear strength, S_{FS}

VARIABLE	DESCRIPTION
S_AB*	Matrix mode shear strength, ab plane, see below, S_{ab}
S_BC*	Matrix mode shear strength, bc plane, see below, S_{bc}
S_CA*	Matrix mode shear strength, ca plane, see below, S_{ca}
SFFC	Scale factor for residual compressive strength, $S_{\it FFC}$
AMODEL	Material models: EQ.1: Unidirectional lamina model EQ.2: Fabric lamina model
PHIC	Coulomb friction angle for matrix and delamination failure, φ < 90°
S_DELM	Scale factor for delamination criterion, S
OMGMX	Limit damage parameter for elastic modulus reduction, ϖ_{\max}
E_LIMT	Element eroding axial strain
ECRSH	Limit compressive relative volume for element eroding
EEXPN	Limit expansive relative volume for element eroding
CERATE1	Coefficient for strain rate dependent strength properties, C_{rate1}
CERATE2	Coefficient for strain rate dependent axial moduli, C_{rate2}
CERATE3	Coefficient for strain rate dependent shear moduli, $C_{\it rate3}$
CERATE4	Coefficient for strain rate dependent transverse moduli, C_{rate4}
AM1	Coefficient for strain softening property for fiber damage in a direction, m_1
AM2	Coefficient for strain softening property for transverse compressive matrix failure mode in b direction (unidirectional) or for fiber damage mode in b direction (fabric), m_2
AM3	Coefficient for strain softening property for fiber crush and punch shear damage, m_3
AM4	Coefficient for strain softening property for matrix failure and delamination damage, m_4

[•] LS-DYNA KEYWORD manual presents these parameters as: SAB, SBC, & SCA. Since SBC is also used to define the "transverse compressive strength," we have used S_xx to represent shear strength in xx plane.

Figure 2.1. from the KEYWORD Manual of LS-DYNA® is reproduced here for the convenience of the MAT162 users.

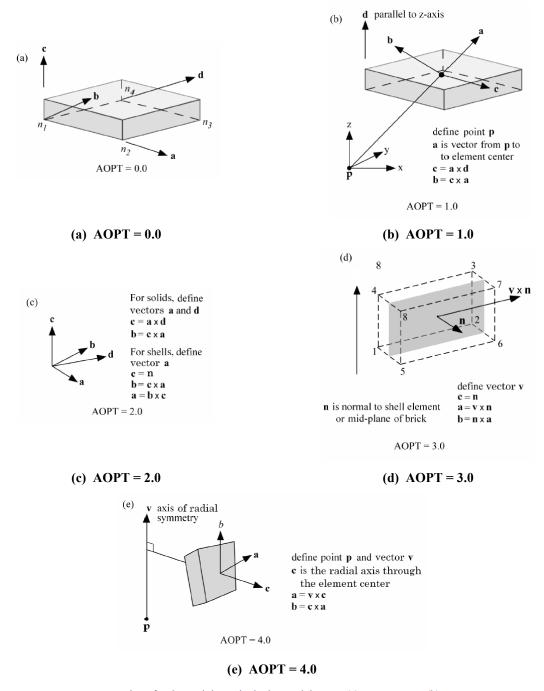


Figure 2.1. Options for determining principal material axes: (a) AOPT = 0.0, (b) AOPT = 1.0 for brick elements, (c) AOPT = 2.0, (d) AOPT = 3.0, and (e) AOPT=4.0 for brick elements.

Figure 1: Material Axes Definition presented in the LS-DYNA KEYWORD Manual (OLDER VERSIONS)

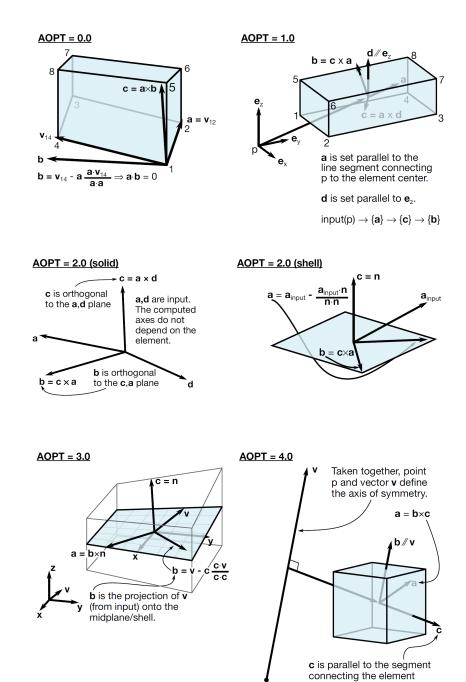


Figure 1 (Continued): Material Axes Definition presented in the LS-DYNA Manuals (NEW VERSIONS, May 2014)

center to the symmetry axis.

MATERIAL MODELS

Failure models based on the 3D stresses/strains in a composite lamina with improved progressive failure modeling capability are established for a unidirectional and for a fabric composite lamina. While the LS-DYNA KEYWORD manual presents the stress based formulations, this manual presents the strain based formulations. These models can be used to effectively simulate the fiber failure, matrix failure, and delamination behavior of composites under high strain-rate and high pressure ballistic impact conditions.

The unidirectional and fabric lamina failure criteria and the associated property degradation models are described as follows. All the failure criteria are expressed in terms of stress components based on ply level strains (ε_1 , ε_2 , ε_3 , ε_{12} , ε_{23} , ε_{31}) = (ε_a , ε_b , ε_c , ε_{ab} , ε_{bc} , ε_{ca}). The associated elastic moduli are (E_1 , E_2 , E_3 , G_{12} , G_{23} , G_{31}) = (E_a , E_b , E_c , G_{ab} , G_{bc} , G_{ca}). Note that for the unidirectional model, a, b, and c denote the fiber, in-plane transverse and out-of-plane or through-thickness directions, respectively; while for the fabric model, a, b, and c denote the in-plane fill, in-plane warp and out-of-plane or through-thickness directions, respectively.

UNIDIRECTIONAL LAMINA DAMAGE MODEL

Fiber Mode Failures

The fiber failure criteria of Hashin [1980] for a unidirectional layer are generalized to characterize the fiber damage in terms of strain components for a unidirectional layer. Three damage functions are used for fiber failure, one in tension/shear, one in compression, and another one in crush under pressure. They are chosen in terms of quadratic strain forms as follows:

UNIDIRECTIONAL TENSION-SHEAR FIBER MODE 1 uf along direction 1, A, a:

$$f_1 - r_1^2 = \left(\frac{E_a \langle \varepsilon_a \rangle}{S_{aT}}\right)^2 + \left(\frac{G_{ab}^2 \varepsilon_{ab}^2 + G_{ca}^2 \varepsilon_{ca}^2}{S_{FS}^2}\right) - r_1^2 = 0 \tag{1}$$

UNIDIRECTIONAL COMPRESSION FIBER MODE 2uf along direction 1, A, a:

$$f_2 - r_2^2 = \left(\frac{E_a \langle \varepsilon_a' \rangle}{S_{aC}}\right)^2 - r_2^2 = 0, \rightarrow \varepsilon_a' = -\varepsilon_a - \frac{\langle -E_c \varepsilon_c - E_b \varepsilon_b \rangle}{2E_a}$$
 (2)

UNIDIRECTIONAL CRUSH FIBER MODE 3uf along direction 3, C, c:

$$f_3 - r_3^2 = \left(\frac{E_c \left\langle -\varepsilon_c \right\rangle}{S_{FC}}\right)^2 - r_3^2 = 0 \tag{3}$$

where $\langle \ \rangle$ are Macaulay brackets, S_{aT} and S_{aC} are the tensile and compressive strengths in the fiber direction, and S_{FS} and S_{FC} are the layer strengths associated with the fiber shear and crush failure, respectively. The damage thresholds, r_j , j=1,2,3, have the initial values equal to 1 before the damage initiated, and are updated due to damage accumulation in the associated damage modes.

Matrix Mode Failures

Matrix mode failures must occur without fiber failure, and hence they will be on planes parallel to fibers. Two matrix damage functions are chosen for the failure plane perpendicular and parallel to the layering planes. They have the forms:

UNIDIRECTIONAL TRANSVERSE COMPRESSION MATRIX MODE 4um along direction 2, B, b:

$$f_4 - r_4^2 = \left(\frac{E_b \left\langle -\varepsilon_b \right\rangle}{S_{bC}}\right)^2 - r_4^2 = 0 \tag{4}$$

UNIDIRECTIONAL PERPENDICULAR MATRIX SHEAR MODE 5um along direction 2, B, b (tension only) and on planes 12, AB, ab & 23, BC, bc:

$$f_5 - r_5^2 = \left(\frac{E_b \langle \varepsilon_b \rangle}{S_{bT}}\right)^2 + \left(\frac{G_{bc} \varepsilon_{bc}}{S_{bc0} + S_{SRB}}\right)^2 + \left(\frac{G_{ab} \varepsilon_{ab}}{S_{ab0} + S_{SRB}}\right)^2 - r_5^2 = 0$$
 (5)

UNIDIRECTIONAL PARALLEL MATRIX MODE (DELAMINATION) 6um along direction 3, C, c (tension only) and on planes 23, BC, bc & 31, CA, ca:

$$f_6 - r_6^2 = S^2 \left\{ \left(\frac{E_c \langle \varepsilon_c \rangle}{S_{cT}} \right)^2 + \left(\frac{G_{bc} \varepsilon_{bc}}{S_{bc0} + S_{SRC}} \right)^2 + \left(\frac{G_{ca} \varepsilon_{ca}}{S_{ca0} + S_{SRC}} \right)^2 \right\} - r_6^2 = 0$$
 (6)

where S_{bT} and S_{cT} are the transverse tensile strengths of the corresponding tensile modes ($\varepsilon_b > 0$ or $\varepsilon_c > 0$); and S_{ab0} , S_{bc0} , & S_{ca0} are the quasi-static shear strength values. Under compressive transverse strain, $\varepsilon_b < 0$ or $\varepsilon_c < 0$, the damaged surface is considered to be "closed", and the shear strengths are assumed to depend on the compressive normal strains based on the Mohr-Coulomb theory, i.e.:

$$S_{SRB} = E_b \tan(\varphi) \langle -\varepsilon_b \rangle$$

$$S_{SRC} = E_c \tan(\varphi) \langle -\varepsilon_c \rangle$$
(7)

where φ is a material constant as $\tan(\varphi)$ is similar to the coefficient of friction. The damage thresholds r_j , j = 4, 5, 6, have the initial values equal to 1 before the damage initiated, and are updated due to damage accumulation of the associated damage modes.

Failure predicted by the criterion of f_4 and f_5 can be referred to as transverse matrix failure, while the matrix failure predicted by f_6 , which is parallel to the layer, can be referred as the delamination mode when it occurs within the elements that are adjacent to the ply interface. Note that a scale factor S is introduced to provide better correlation of delamination area with experiments. The scale factor S can be determined by fitting the analytical prediction to experimental data for the delamination area.

FABRIC LAMINA DAMAGE MODEL

Fiber Mode Failures

The fiber failure criteria of Hashin [1980] for a unidirectional layer are generalized to characterize the fiber damage in terms of strain components for a plain weave layer. The fill and warp fiber tensile/shear damage are given by the quadratic interaction between the associated axial and through the thickness shear strains, i.e.:

TENSION-SHEAR FIBER MODE 7f along direction 1, A, a (tension only), and on plane 31, CA, ca [Eq. 8a]:

TENSION-SHEAR FIBER MODE 8f along direction 2, B, b (tension only), and on plane 23, BC, bc [Eq. 8b]:

$$f_7 - r_7^2 = \left(\frac{E_a \langle \varepsilon_a \rangle}{S_{aT}}\right)^2 + \left(\frac{G_{ca}\varepsilon_{ca}}{S_{aFS}}\right)^2 - r_7^2 = 0$$
 (8a)

$$f_8 - r_8^2 = \left(\frac{E_b \langle \varepsilon_b \rangle}{S_{bT}}\right)^2 + \left(\frac{G_{bc} \varepsilon_{bc}}{S_{bFS}}\right)^2 - r_8^2 = 0$$
 (8b)

where S_{aT} and S_{bT} are the axial tensile strengths in the fill and warp directions, respectively, and S_{aFS} and S_{bFS} are the lamina shear strengths due to fiber shear failure in the fill and warp directions. These failure criteria are applicable when the associated ε_a or ε_b is positive. The damage thresholds r_7 and r_8 are equal to 1 without damage. It is assumed $S_{aFS} = S_{FS}$, and $S_{bFS} = S_{FS} \times S_{bT} / S_{aT}$.

COMPRESSION FIBER MODE 9f along direction 1, A, a [Eq. 9a]: COMPRESSION FIBER MODE 10f along direction 2, B, b [Eq. 9b]:

When ε_a or ε_b is compressive, it is assumed that the in-plane compressive damage in the fill and warp directions are given by the maximum strain criterion, i.e.:

$$f_9 - r_9^2 = \left(\frac{E_a \left\langle \varepsilon_a' \right\rangle}{S_{aC}}\right)^2 - r_9^2 = 0 \quad \rightarrow \quad \varepsilon_a' = -\varepsilon_a - \left\langle -\varepsilon_c \right\rangle \frac{E_c}{E_a} \tag{9a}$$

$$f_{10} - r_{10}^2 = \left(\frac{E_b \left\langle \varepsilon_b' \right\rangle}{S_{bC}}\right)^2 - r_{10}^2 = 0 \quad \rightarrow \quad \varepsilon_b' = -\varepsilon_b - \left\langle -\varepsilon_c \right\rangle \frac{E_c}{E_b} \tag{9b}$$

where S_{aC} and S_{bC} are the axial compressive strengths in the fill and warp directions, respectively, and r_{θ} and $r_{1\theta}$ are the corresponding damage thresholds. Note that the effect of through the thickness compressive strain on the in-plane compressive damage is taken into account in the above two equations.

CRUSH FIBER MODE 11f along direction 3, C, c [Eq. 10]:

When a composite material is subjected to transverse impact by a projectile, high compressive stresses will generally occur in the impact area with high shear stresses in the surrounding area between the projectile and the target material. While the fiber shear punch damage due to the high shear stresses can be accounted for by equation (1), the crush damage due to the high through the thickness compressive pressure is modeled using the following criterion:

$$f_{11} - r_{11}^2 = \left(\frac{E_c \left\langle -\varepsilon_c \right\rangle}{S_{FC}}\right)^2 - r_{11}^2 = 0 \tag{10}$$

where S_{FC} is the fiber crush strengths and r_{II} is the associated damage threshold.

Matrix Mode Failures

IN-PLANE MATRIX SHEAR MODE 12m on plane 12, AB, ab [Eq. 11]:

A plain weave layer can be damaged under in-plane shear stressing without occurrence of fiber breakage. This in-plane matrix damage mode is given by:

$$f_{12} - r_{12}^2 = \left(\frac{G_{ab}\varepsilon_{ab}}{S_{ab}}\right)^2 - r_{12}^2 = 0 \tag{11}$$

where S_{ab} is the layer shear strength due to matrix shear failure and r_{12} is the damage threshold.

PARALLEL MATRIX SHEAR & DELAMINATION MODE 13m along direction 3, C, c (tension only) and on planes 23, BC, bc & 31, CA, ca [Eq. 12]:

Another failure mode, which is due to the quadratic interaction between the transverse strains, is expected to be mainly a matrix failure. This through the thickness matrix failure criterion is assumed to have the following form:

$$f_{13} - r_{13}^2 = S^2 \left\{ \left(\frac{E_c \langle \varepsilon_c \rangle}{S_{cT}} \right)^2 + \left(\frac{G_{bc} \varepsilon_{bc}}{S_{bc0} + S_{SRC}} \right)^2 + \left(\frac{G_{ca} \varepsilon_{ca}}{S_{ca0} + S_{SRC}} \right)^2 \right\} - r_{13}^2 = 0$$
 (12)

where r_{13} is the damage threshold, S_{cT} is the through the thickness tensile strength for tensile ε_c , and $S_{bc\theta}$ and $S_{ca\theta}$ are the quasi-static shear strengths. The damage surface due to equation (12) is parallel to the composite layering plane. Under compressive through the thickness strain, $\varepsilon_c < 0$, the damaged surface (delamination) is considered to be "closed", and the shear strengths are assumed to depend on the compressive normal strain ε_c similar to the Mohr-Coulomb theory, i.e.:

$$S_{SRC} = E_c \tan(\varphi) \langle -\varepsilon_c \rangle \tag{13}$$

where φ is the Coulomb's friction angle. When damage predicted by this criterion occurs within elements that are adjacent to the ply interface, the failure plane is expected to be parallel to the layering planes, and, thus, can be referred to as the delamination mode. Note that a scale factor S is introduced to provide better correlation of delamination area with experiments. The scale factor S can be determined by fitting the analytical prediction to experimental data for the delamination area.

DAMAGE PROGRESSION MODEL

A set of damage variables ϖ_i with i = 1, ..., 6; are introduced to relate the onset and growth of damage to stiffness losses in the material. The compliance matrix [S] is related to the damage variables as (Matzenmiller, et al., 1995):

$$[S] = \begin{bmatrix} \frac{1}{(1-\varpi_1)E_a} & \frac{-\nu_{ba}}{E_b} & \frac{-\nu_{ca}}{E_c} & 0 & 0 & 0\\ \frac{-\nu_{ab}}{E_a} & \frac{1}{(1-\varpi_2)E_b} & \frac{-\nu_{cb}}{E_c} & 0 & 0 & 0\\ \frac{-\nu_{ac}}{E_a} & \frac{-\nu_{bc}}{E_b} & \frac{1}{(1-\varpi_3)E_c} & 0 & 0 & 0\\ 0 & 0 & 0 & \frac{1}{(1-\varpi_4)G_{ab}} & 0 & 0\\ 0 & 0 & 0 & 0 & \frac{1}{(1-\varpi_5)G_{bc}} & 0\\ 0 & 0 & 0 & 0 & 0 & \frac{1}{(1-\varpi_5)G_{bc}} \end{bmatrix}$$

$$(14)$$

The stiffness matrix [C] is obtained by inverting the compliance matrix, i.e., $[C] = [S]^{-1}$. As suggested in Matzenmiller, et al., (1995), the growth rate of damage variables, ϖ_i , is governed by the damage rule of the form:

$$\dot{\boldsymbol{\sigma}}_{i} = \max\left\{\dot{\boldsymbol{\phi}}_{i} q_{ij}\right\} \tag{15}$$

where the scalar damage functions $\dot{\phi}_j$ control the amount of growth and the vector-valued matrix q_{ij} (i = 1,...6, j = 1, ..., 13) provide the coupling between the individual damage variables (i) and the various damage modes (j). Note that there are six damage modes (j = 1, ..., 6) for the "unidirectional lamina model" and seven damage modes (j = 7, ..., 13) for the "fabric lamina model." The damage criteria $f_j - r_j^2 = 0$ of Eqs. (1) to (12) provide the damage surfaces in strain space for the unidirectional and fabric lamina models, respectively. Damage growth, $\dot{\phi}_j > 0$, will occur when the strain path crosses the updated damage surface $f_j - r_j^2 = 0$ and the strain increment has a non-zero component in the direction of the normal to the damage surface, i.e., $\sum_i \frac{\partial f_j}{\partial \varepsilon_i} \dot{\varepsilon}_i > 0$. Combined with damage growth functions $\gamma_j \left(\varepsilon_i, \varpi_j\right)$; $\dot{\phi}_j$ is assumed to have the form:

$$\dot{\phi}_{j} = \sum_{i} \gamma_{j} \frac{\partial f_{j}}{\partial \varepsilon_{i}} \dot{\varepsilon}_{i} \quad \text{(no summation over j)}$$
 (16)

Choosing

$$\gamma_{j} = \frac{1}{2} \left(1 - \phi_{j} \right) f_{j}^{\frac{m_{j}}{2} - 1} \tag{17}$$

and noting that

$$\sum_{i} \frac{\partial f_{j}}{\partial \varepsilon_{i}} \dot{\varepsilon}_{i} = \dot{f}_{j} \tag{18}$$

for the quadratic functions given by Eqs. (1) to (6) and Eqs. (8) to (12), lead to:

$$\dot{\phi}_{j} = \frac{1}{2} \left(1 - \phi_{j} \right) f_{j}^{\frac{m_{j}}{2} - 1} \dot{f}_{j} \tag{19}$$

where ϕ_j is the scalar damage function associated with the jth failure mode, and m_j is a material constant for softening behavior. The scalar damage function ϕ_j can be obtained by integrating Eq. (19) as follows:

$$\dot{f}_{j} = 2r_{j}\dot{r}_{j} & \& \\
f_{j}^{m_{j}} = r_{j}^{m_{j}-2} = r_{j}^{m_{j}-2} \\
\dot{\phi}_{j} = (1 - \phi_{j})r_{j}^{m_{j}-1}\dot{r}_{j} & \rightarrow \int_{0}^{\phi_{j}} \frac{d\phi_{j}}{(1 - \phi_{j})} = \int_{0}^{r_{j}} r_{j}^{m_{j}-1}dr_{j} \dots \rightarrow \phi_{j} = 1 - \exp\left(\frac{1}{m_{j}}(1 - r_{j}^{m_{j}})\right)$$
(20)

The damage coupling matrix q_{ij} is considered for the unidirectional and fabric lamina models as follows.

DAMAGE COUPLING MATRIX FOR UNIDIRECTIONAL LAMINA MODEL

Eq. (21) is the damage coupling matrix, and Fig. 2 illustrates how Eq. (21) is associated with the modulus reduction for the unidirectional lamina model.

$$q_{ij}^{U} = \begin{bmatrix} 1 & 1 & 1 & 0 & 0 & 0 \\ 0 & 0 & 1 & 1 & 1 & 0 \\ 0 & 0 & 1 & 0 & 0 & 1 \\ 1 & 1 & 1 & 1 & 1 & 0 \\ 0 & 0 & 1 & 1 & 1 & 1 \\ 1 & 1 & 1 & 0 & 0 & 1 \end{bmatrix} \quad i = 1, ..., 6; \ j = 1, ..., 6.$$

$$(21)$$

	MAGE PES	FIBER	DAMAGE M	IODES	MATRIX DAMAGE MODES			
	MAGE DES	MODE 1uf j = 1	MODE 2uf j = 2	MODE 3uf j = 3	MODE 4um j = 4	MODE 5um j = 5	MODE 6um j = 6	
MODULI	q_{ij}^U							
Ea		1	1	1	0	0	0	
E _b		0	0	1	1	1	0	
Ec		0	0	1	0	0	1	
Gab		1	1	1	1	1	0	
Gbc		0	0	1	1	1	1	
Gca		1	1	1	0	0	1	

Figure 2: Coupling of Different Damage Modes to the Associated Reduction in Moduli for Unidirectional Lamina Model.

DAMAGE COUPLING MATRIX FOR FABRIC LAMINA MODEL

Eq. (22) is the damage coupling matrix, and Fig. 3 illustrates how Eq. (22) is associated with the modulus reduction for the fabric lamina model.

$$q_{ij}^{F} = \begin{bmatrix} 1 & 0 & 1 & 0 & 1 & 0 & 0 \\ 0 & 1 & 0 & 1 & 1 & 0 & 0 \\ 0 & 0 & 0 & 0 & 1 & 0 & 1 \\ 1 & 1 & 1 & 1 & 1 & 1 & 0 \\ 0 & 1 & 0 & 1 & 1 & 0 & 1 \\ 1 & 0 & 1 & 0 & 1 & 0 & 1 \end{bmatrix}$$
 $i = 1, ..., 6; j = 7, ..., 13.$ (22)

	DAMAGE YPES		FIBER	MATRIX DAMAGE MODES				
PW DAMAGE MODES		MODE 7f j = 7	MODE 8f j = 8	MODE 9f j = 9	MODE 10f j = 10	MODE 11f j = 11	MODE 12m j = 12	MODE 13 m j = 13
MO -DU -LI	q^F_{ij}							
Ea		1	0	1	0	1	0	0
Еь		0	1	0	1	1	0	0
Ec		0	0	0	0	1	0	1
Gab		1	1	1	1	1	1	0
Gbc		0	1	0	1	1	0	1
Gca		1	0	1	0	1	0	1

Figure 3: Coupling of Different Damage Modes to the Associated Reduction in Moduli for Fabric Lamina Model.

Through Eq. (15), the damage coupling matrix q_{ij} relates the individual damage variables ϖ_i to the various damage modes provided by the scalar damage functions ϕ_j for the unidirectional and fabric lamina models.

Unidirectional Fiber Modes 1uf, 2uf, & 3uf: For the unidirectional lamina model, the damage coupling vectors q_{il} and q_{i2} of equation (21) are chosen such that the fiber tension-shear and compressive damage modes 1uf and 2uf, Eqs. (1) & (2), provide the reduction of elastic moduli E_a , G_{ab} , and G_{ca} , due to ϖ_l , ϖ_d and ϖ_b , respectively. The coupling vector q_{i3} provides that all the elastic moduli are reduced due to the fiber crush damage mode 3uf, Eq. (3).

Unidirectional Matrix Modes 3um, 4um, & 5um: For the transverse matrix damage modes 4um and 5um, Eqs. (4) & (5), q_{i4} and q_{i5} provide the reduction of E_b , G_{ab} and G_{bc} , while for the through thickness matrix damage mode 6um, q_{i6} provides the reduction of E_c , G_{bc} , and G_{ca} .

Fabric Fiber Modes 7f, 8f, 9f, 10f, & 11f: For the fabric lamina model, the damage coupling vectors q_{i7} , q_{i8} , q_{i9} and q_{i10} are chosen for the fiber tension-shear and compressive damage modes 7f to 10f, Eqs. (8), & (9); such that the fiber damage in either the fill or warp direction results in stiffness reduction in the loading direction and in the related shear directions. For the fiber crush damage mode 11f, Eq (10), the damage coupling vector q_{i11} is chosen such that all the stiffness values are reduced as an element is failed under the crush mode.

Fabric Matrix Modes 12m, & 13m: For the in-plane matrix shear failure mode 12m given by Eq. (11), the stiffness reduction due to q_{il2} is limited to in-plane shear modulus, while the through thickness matrix damage (delamination) mode 13m, the coupling vector q_{il3} is chosen for the through thickness tensile modulus and transverse shear moduli.

NON-LINEAR PROGRESSIVE DAMAGE MODEL OF MAT162

Utilizing the damage coupling matrix given by Eqs. (21) & (22), and the scalar damage function given by Eq. (20), the damage variables ϖ_i can be obtained from Eq. (15) for an individual failure mode j as:

$$\varpi_i = 1 - \exp\left\{\frac{1}{m_i} \left(1 - r_i^{m_j}\right)\right\} \qquad \qquad r_j \ge 1 \tag{23}$$

Note that the damage thresholds r_j given in the damage criteria of Eqs. (1) to (12) are continuously increasing functions with increasing damage. The damage thresholds have an initial value of one, which results in a zero value for the associated damage variable ϖ_i from Eq. (23). This provides an initial elastic region bounded by the damage functions in strain space. The nonlinear response is modeled by loading on the damage surfaces to cause damage growth with increasing damage thresholds and the values of damage variables ϖ_i . After damage initiated, the progressive damage model assumes linear elastic response within the part of strain space bounded by the updated damage thresholds. The elastic response is governed by the reduced stiffness matrix associated with the updated damage variables ϖ_i given in Eq. (14).

In defining the non-linear stress-strain behavior of a composite material in a specific direction k, a damage threshold r_k (k = 1, ..., 6) can also be expressed as the ratio between the current total strain in the kth direction and the corresponding yield strain.

$$r_k = \frac{\varepsilon_k}{\varepsilon_{kv}} \tag{24}$$

From Eq. 14, the Young's modulus in the kth direction can now be expressed as:

$$E_k = (1 - \omega_k) E_{k0} = E_{k0} \exp\left\{\frac{1}{m_i} \left(1 - \left(\frac{\varepsilon_k}{\varepsilon_{ky}}\right)^{m_k}\right)\right\}$$
 (25)

Since the reduced modulus is also considered linear, the stress-strain relationship of the damaged material can now be expressed as:

$$\sigma_k = E_k \varepsilon_k = E_{k0} \varepsilon_k \exp\left\{ \frac{1}{m_j} \left(1 - \left(\frac{\varepsilon_k}{\varepsilon_{ky}} \right)^{m_k} \right) \right\}$$
 (26)

Recognizing the fact that $\sigma_{ky} = E_{k0} \varepsilon_{ky}$ Eq. (26) can also be expressed as:

$$\frac{\sigma_k}{\sigma_{ky}} = \frac{\varepsilon_k}{\varepsilon_{ky}} \exp\left\{ \frac{1}{m_j} \left(1 - \left(\frac{\varepsilon_k}{\varepsilon_{ky}} \right)^{m_k} \right) \right\} \tag{27}$$

Fig. 4 shows the plot of Eq. (27) for different values of damage softening parameter, m. Note that the value of $r = \varepsilon / \varepsilon_y \le 1$, represent the linear-elastic part of the stress-strain behavior, and Eq. (27) represents the post-yield damage softening behavior for $r = \varepsilon / \varepsilon_y \ge 1$.

It is well known that it is difficult to obtain the softening response of most quasi-brittle materials including fiber-reinforced composites. The softening response heavily depends on the set-up and test machines, which can lead to very scattered results. Consequently the choice of damage parameters for each mode becomes an open issue. Generally, smaller values of *m* make the material more ductile whereas higher values give the material more brittle behavior. A methodology to systematically determine the model material properties for penetration modeling has been successfully established in [Xiao et al., 2005].

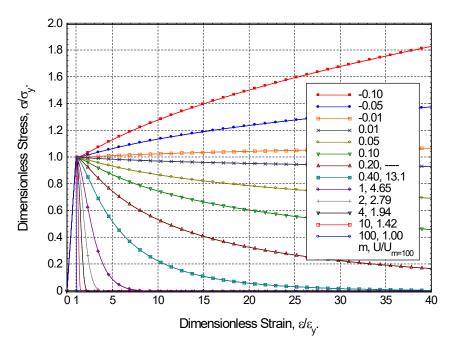


Figure 4: Non-Linear Progressive Damage Model of MAT162. Post-Yield Damage Softening of a Composite as a Function of Damage Softening Parameter m.

In MAT 162, the damage softening parameter m_1 controls the tensile and compressive fiber failure mode in a direction, and m_2 controls the transverse compressive matrix failure mode in b direction for the "unidirectional lamina model." However, for fabric the "fabric lamina model," m_2 controls the softening of tensile and compressive fiber failure mode in b direction. m_3 is for softening related to fiber crush mode, and m_4 is for both perpendicular and parallel matrix mode for "unidirectional" case, and for both in-plane matrix failure and through the thickness matrix failure for "fabric" case. Detail analysis on m parameters on the stress-strain behavior can be found in Ref. [Gama et al., 2009].

ADDITIONAL DISCUSSION ON MODULII AND STRENGTH REDUCTION

When fiber tension-shear damage is predicted in a layer by equation (1) or (8), the load carrying capacity of that layer in the associated direction is reduced to zero according to damage variable Eq. (23). For compressive fiber damage due to equation (2) or (9), the layer is assumed to carry a residual axial load in the damaged direction. The damage variables of Eq. (23) for the compressive modes have been modified to account for the residual strengths of $S_{aCR} = S_{aC} \times S_{FFC}$ and $S_{bCR} = S_{bC} \times S_{FFC}$ in the fill and warp directions, respectively.

For through thickness matrix (delamination) failure given by equation (6) or (12), the in-plane load carrying capacity within the element is assumed to be elastic (i.e., no in-plane damage). The load carrying behavior in the through thickness direction is assumed to depend on the opening or closing of the matrix damage surface. For tensile mode, $\varepsilon_c > 0$, the through thickness

stress components are softened and reduced to zero due to the damage criteria described above. For compressive mode, $\varepsilon_c < 0$, the damage surface is considered to be closed, and thus, ε_c is assumed to be elastic, while ε_{bc} and ε_{ca} are allowed to reduce to sliding friction traction of equation (7) or (13). Accordingly, for the through thickness matrix failure under compressive mode, the damage variable equation is further modified such that the residual sliding strength value is equal to S_{SRC} .

EFFECT OF STRAIN RATES ON STRENGTH AND MODULI

The effect of strain-rate on the nonlinear stress-strain response of a composite layer is modeled by a logarithmic strain-rate dependent function for the moduli and strength of the form:

$$\frac{X_{RT}}{X_0} = 1 + C_{rate} \ln \left(\frac{\dot{\overline{\varepsilon}}}{\dot{\overline{\varepsilon}}_0} \right)$$
 (28)

where, X_{RT} is the rate dependent property of interest at an average strain rate of $\dot{\bar{\varepsilon}}$, and X_0 is the quasi-static property of interest at an average reference strain rate of $\dot{\bar{\varepsilon}}_0$. In the present MAT162 formulation, the reference strain rate is chosen to be:

$$\dot{\bar{\varepsilon}}_0 = 1$$
 per time unit (29)

If the time unit is sec, the reference strain rate is 1/s. Similarly, if the time unit is μs , the the reference strain is $1/\mu s$.

EFFECT OF STRAIN RATE ON STRENGTH PROPERTIES

One average rate parameter, CERATE1 or C_{rate1} is used to add rate effects on strength properties as follows:

$$\{S_{RT}\} = \{S_0\} \left(1 + C_{rate1} \ln \left(\frac{\{\dot{\bar{\varepsilon}}\}}{\dot{\bar{\varepsilon}}_0} \right) \right)$$
 (30)

where, the strength and strain rate matrices are given by Eq. (31). Note that the through thickness tensile strength S_{cT} , and the shear strengths S_{ab} , S_{bc} , & S_{ca} ; are not considered as rate dependent in MAT162 formulation.

EFFECT OF STRAIN RATE ON MODULI

Three rate parameters, C_{rate2} , C_{rate3} , & C_{rate4} are used to add rate effects on three axial and three shear moduli as presented in Eq. (32), where, the moduli, strain rate, and rate parameter matrices

are given by Eq. (33). Note that the rate effects on both the axial moduli, $E_a \& E_b$, are controlled by the rate parameter C_{rate2} , and that for the through thickness modulus, E_c , by C_{rate4} . In addition, the rate effects on the shear moduli, G_{ab} , G_{bc} , & G_{ca} , are controlled by the rate parameter C_{rate3} .

$$\{S\} = \begin{cases}
S_{aT} \\
S_{aC} \\
S_{bT} \\
S_{bC} \\
S_{FC} \\
S_{FS}
\end{cases}, & \{\dot{\bar{\varepsilon}}\} = \begin{cases}
|\dot{\varepsilon}_{a}| \\
|\dot{\varepsilon}_{a}| \\
|\dot{\varepsilon}_{b}| \\
|\dot{\varepsilon}_{b}| \\
|\dot{\varepsilon}_{c}| \\
(\dot{\varepsilon}_{ca}^{2} + \dot{\varepsilon}_{bc}^{2})^{1/2}
\end{cases}$$
(31)

$$\{E_{RT}\} = \{E_0\} \left(1 + \{C_{rate}\} \ln \frac{\{\dot{\overline{\varepsilon}}\}}{\dot{\varepsilon}_0}\right)$$
(32)

A discussion of rate dependent MAT162 properties can be found in Ref. [Gama & Gillespie Jr. 2011].

ELEMENT EROSION

A failed element is eroded in any of three different ways:

- 1. If fiber tensile failure in a "unidirectional" layer is predicted in the element and the axial tensile strain is greater than E_LIMIT. For a "fabric" layer, both in-plane directions are failed and exceed E_LIMIT.
- 2. If compressive relative volume (ratio of current volume to initial volume) in a failed element is smaller than ECRSH.
- 3. If expansive relative volume in a failed element is greater than EEXPN.

FINITE ELEMENT MODELING TIPS

- One point integration solid element (TYPE = 1) can be used for MAT162.
- In order to observe the delamination at the interface between two adjacent laminas, two different PART IDs with different MAT IDs for each parts and with different material orientation angles (BETA in the MAT162 cards) must be defined at the interface of interest. If it is required to model the delamination at the interface between two plies with the same material orientation angles, those two angles must be defined in different ways in each PART, e.g., BETA = 0.00 & BETA = 180.00.
- *DATABASE EXTENT BINARY must be included to check history variables.
- Type of *HOURGLASS need to be checked for minimum hourglass energy over the duration of the LS-DYNA solution.

DAMAGE HISTORY PARAMETERS

Information about the damage history variables for the associated failure modes can be plotted in LS-POST. These additional variables are tabulated below:

History Variable			Description	Value	LS-POST
#	Uni	Fabric	Description	value	Components
1	$\max(r_1, r_2)$	$\operatorname{Max}(r_7, r_9)$	Fiber mode in a		7
2	-	Max (r_8, r_{10})	Fiber mode in b		8
3	r_3	r_{II}	Fiber crush mode	Fiber crush mode 0 - elastic	
4	r ₅	r_{12}	Perpendicular matrix mode	> 1- damage thresholds, Eqs. (1-6) to (8-12)	10
5	r_6	r_{I3}	Parallel matrix/ delamination mode		11
	6		Element delamination indicator	0 – no delamination 1 – with delamination	12

In LS-PREPOST, history variables of an element can be plotted from [History] tab. History var#7 to history var#12 are used for MAT162 damage variables. History variables can also be viewed as fringe component tab [FrinComp]. It is important to know that, in visualizing the MAT162 damage modes, the fringe range in the [FriRang] tab need to set to [User], the minimum and maximum value of the [User] range shall be set to [Min: = 0] & [Max: = 1] and finally the [Update] & [Done] buttons need to be pressed sequentially. The table above presents the relationship between damage thresholds, r_j , with the history variables for both "Unidirectional" and "Fabric" lamina models, and are presented separately in the following tables.

Damage Mode	Mode No.	Equation	Eq. No.	Mat. Dir.	LS- <u>PrePost</u> HISV No.	HISV Values
Tension-Shear Fiber Mode	luf		(1)	1	$7 \\ [\max\{r_1, r_2\}]$	*
Compression Fiber Mode	2uf	$f_2 - r_2^2 = \left(\frac{E_a \left\langle \varepsilon_a' \right\rangle}{S_{aC}}\right)^2 - r_2^2 = 0$	(2)	1	$7 \\ [\max\{r_1, r_2\}]$	ak:
Crush Fiber Mode 3	3uf		(3)	3	9 [r ₃]	*
Transverse Compression Matrix Mode	4um	$f_4 - r_4^2 = \left(\frac{E_b \left\langle -\varepsilon_b \right\rangle}{S_{bC}}\right)^2 - r_4^2 = 0$	(4)	2	8 [r ₄]	*
Perpendicular Matrix Shear Mode	5um		(5)	12	10 [r ₅]	s c
Parallel Matrix Shear Delamination Mode	6um		(6)	13	11 [r ₆]	ης
Delamination Index					12 [r ₆]	* **

*	0 – elastic	> 1 – damage threshold [Equations (1) to (6)]
*	0 – no delamination	1 – full delamination

For the unidirectional damage modes, the damage coupling matrix [also presented below] given by Eq. (21) shows that for one specific damage mode, several elastic constants will be degraded.

- Tension-Shear Fiber Mode 1uf [TSFM-1uf]: $[E_A, G_{AB}, G_{CA}]$ will be degraded
- Compression Fiber Mode 2uf [CompFM-2uf]: $[E_A, G_{AB}, G_{CA}]$ will be degraded
- Crush Fiber Mode 3uf [CrshFM-3uf]: $[E_A, E_B, E_C, G_{AB}, G_{BC}, G_{CA}]$ will be degraded
- Trans. Comp. Matrix Mode 4um [TCMM-4um]: $[E_B, G_{AB}, G_{BC}]$ will be degraded
- Para. Matrix Shear Mode 5um [PMSM-5um]: $[E_B, G_{AB}, G_{BC}]$ will be degraded
- Perp. Matrix Shear Delam. Mode 6um [PMSDM-6um]: $[E_C, G_{BC}, G_{CA}]$ will be degraded

$\mathbf{q_{ij}^U}$	TSFM-1uf	CompFM-2uf	CrshFM-3uf	TCMM-4um	PMSM-5um	PMSDM-6um
E _A	1	1	1	0	0	0
$\mathbf{E}_{\mathbf{B}}$	0	0	1	1	1	0
E _C	0	0	1	0	0	1
G_{AB}	1	1	1	1	1	0
G_{BC}	0	0	1	1	1	1
G_{CA}	1	1	1	0	0	1

Thus, for unidirectional lamina model, each damage mode will degrade three (3) moduli except for crush (6 moduli) mode. Hence loading in one direction may show more than one damage history variable activated.

1	Fabric Lan	nina I	Damage Modes and Corresponding History Varia	ıbles	;
	Damage	Mode	Equation	Eq.	N

Damage Mode	Mode No.	Equation	Eq. No.	Mat. Dir.	LS- <u>PrePost</u> HISV No.	HISV Values
Tension- Shear Fiber Mode 7	7f		(8a)	1	$7 \\ [\max\{r_7, r_9\}]$	*
Tension- Shear Fiber Mode 8	8f	$f_8 - r_8^2 = \left(\frac{E_b \left\langle \mathcal{E}_b \right\rangle}{S_{bT}}\right)^2 + \left(\frac{G_{bc} \mathcal{E}_{bc}}{S_{bFS}}\right)^2 - r_8^2 = 0$	(8b)	2	8 [max $\{r_8, r_{10}\}$]	*
Compression Fiber Mode 9	9f	$\int_{9} -r_{9}^{2} = \left(rac{E_{a}\left\langle \mathcal{E}_{a}^{\prime} ight angle}{S_{aC}} ight)^{2} -r_{9}^{2} = 0$	(9a)	1	$7 \\ [\max\{r_7, r_9\}]$	*
Compression Fiber Mode 10	10f	$f_{10} - r_{10}^2 = \left(\frac{E_b \left\langle \mathcal{E}_b^{\prime} \right\rangle}{S_{bC}}\right)^2 - r_{10}^2 = 0$	(9b)	2	$[\max\{r_8, r_{10}\}]$	*
Crush Fiber Mode 11	11f	$\left f_{11} - r_{11}^2 = \left(\frac{E_c \left\langle -\varepsilon_c \right\rangle}{S_{FC}} \right)^2 - r_{11}^2 = 0 \right $	(10)	3	9 [r ₁₁]	*
In-Plane Matrix Shear Mode 12	12m	$f_{12} - r_{12}^2 = \left(\frac{G_{ab}\varepsilon_{ab}}{S_{ab}}\right)^2 - r_{12}^2 = 0$	(11)	12	10 [r ₁₂]	*
Parallel Matrix Shear Delamination Mode 13	13m		(12)	13 & 23	11 [r ₁₃]	*
Delamination Index					12 [r ₁₃]	*

*	0 – elastic	> 1 – damage threshold [Equations (1) to (6)]
*	0 – no delamination	1 – full delamination

For the fabric damage modes, the damage coupling matrix [also presented below] given by Eq. (22) shows that for one specific damage mode, several elastic constants will be degraded.

- Tension-Shear Fiber Mode 7f [TSFM-7f]: $[E_A, G_{AB}, G_{CA}]$ will be degraded
- Compression Fiber Mode 9f [CompFM-9f]: $[E_A, G_{AB}, G_{CA}]$ will be degraded
- Tension-Shear Fiber Mode 8f [TSFM-8f]: $[E_B, G_{AB}, G_{BC}]$ will be degraded
- Compression Fiber Mode 10f [CompFM-10f]: $[E_B, G_{AB}, G_{BC}]$ will be degraded
- Crush Fiber Mode 11f [CrshFM-11f]: $[E_A, E_B, E_C, G_{AB}, G_{BC}, G_{CA}]$ will be degraded
- In-Plane Matrix Shear Mode 12m [IPMSM-11m]: $[G_{AB}]$ will be degraded
- Perp. Matrix Shear Delam. Mode 12m [PMSDM-6um]: $[E_C, G_{BC}, G_{CA}]$ will be degraded

$\mathbf{q}_{ij}^{\mathrm{U}}$	TSFM-7f	TSFM-8f	CompFM- 9f	CompFM- 10f	CrshFM- 11f	IPMSM- 12m	PMSDM- 13m
E _A	1	0	1	0	1	0	0
E _B	0	1	0	1	1	0	0
E _C	0	0	0	0	1	0	1
G_{AB}	1	1	1	1	1	0	0
G_{BC}	0	1	0	1	1	1	1
G_{CA}	1	0	1	0	1	0	1

Thus each damage mode will degrade three (3) moduli except for crush (6 moduli) and in-plane shear (1 modulus). Hence loading in one direction may show more than one damage history variable activated.

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APPENDIX A: MAT162 PROPERTIES AND PARAMETERS

MAT162 ELASTIC AND STRENGTH PROPERTIES

In addition to ASTM standard test methods, UD-CCM has developed non-standard experimental techniques and computational methodologies to determine all material properties and parameters needed for MAT162.

MAT162 ELASTIC & STRENGTH PROPERTIES: ASTM STANDARD TESTS

Properties, Parameters	Test Method	Specimen	Dimensions (mm)	Miscellaneous
E_1, ν_{12}, X_1^T	0° Tension (ASTM D3039)		254×25.4×1	
E_2, ν_{21}, X_2^T	90° Tension (ASTM D3039)		175×25.4×2	
X_1^C	0° Compression (ASTM D3410)		155×25.4×2	
X_2^C	90° Compression (ASTM D3410)		155×25.4×2	
$E_3, v_{31}, v_{32}, X_3^T$	Thru-thickness Tension/ Compression (no standard)		20×20×20 15×15×30	
$G_{12}, G_{23}, G_{31} \\ S_{12}, S_{23}, S_{31}$	In-Plane +/-45 Tension, Rail-Shear & V-Notch Shear (ASTM D5379)		76×4.5×20	Shear in 1-2, 2-3, 3-1 planes

MAT162 PROPERTIES & PARAMETERS: NON-STANDARD UD-CCM TEST & COMPUTATIONAL METHODOLOGIES

	CB CCM IEST	& COMI UTATIONA	LE TVIE III	ODOLOGILO
E ₃ ,PHIC	Out-of-Plane Off-Axis Compression (no standard)		15×15×15	θ = 0°, 15°, 30°, 45°, 60°, 75°
SFFC	Open Hole Compact Compression Test (no standard)	T W	25×25×13 D = 8~12	Loading along fiber direction for both UD and PW Composites
SFC,SFS	Quasi-Static Punch Shear Test (QS-PST) Punch Crush Strength (PCS) & Punch Shear Strength (PSS)	Punch Head Upper Support Ring Connection Cover Flatte Cover Support Ring User Support Ring Support Ring Diameter	25 mm Discs 100x100 Plates 150x150 Plates	$SPR = D_S/D_P$ $SPR = 0 \text{ for SFC}$ $SPR = 1.1 \text{ for SFS}$
OMGMX & Damage Softening Parameters m ₁ to m ₄	Low Velocity Impact Experiments; & Numerical Simulation (BZH Methodology)	Top Plate Hemispherical Tup Composite Specimen Vertical Suppo Plate	100x150 Plates	Energy Levels: 30J to 70J @ an increment of 10J
Erosion Parameters E_LIMT $EEXPN$ Rate Parameters C_{rate1} to C_{rate4}	Hopkinson Bar Testing Dynamic Punch Shear Ballistic Testing Numerical Simulation of Ballistic Tests (BZH Methodology)	V ₀ † D _P	150x150x15 Plates	
Crush Parameters, SFC, ECRSH	Depth of Penetration Impact Experiments; & Numerical Simulation (BZH Methodology)	Cover Plate Support Leg	305x205x50 Plates	Impact at 300 to 800 m/s @ an interval of 50 m/s. Measure DoP as a function of impact velocity. Parametrically determine SFC & ECRSH to match the Experimental Data.

MAT162 DATABASE OF COMPOSITE PROPERTIES

Properties, Unit	UD S-2 Glass/SC15	PW S-2 Glass/SC15	PW S- Glass/Phenolic OWENS Corning	PW E- Glass/Phenolic U.S. AERDC
\mathbf{v}_{f}	0.60	0.53	0.62	0.66
$ ho_{\scriptscriptstyle C}$, g/cm3	1.85	1.85	2.00	2.107
E1, GPa	64.0	27.5	38.6	29.15
E2, GPa	11.8	27.5	31.9	29.15
E3, GPa	11.8	11.8	12.0	11.00
ν21	0.0535	0.110	0.100	0.078
ν31	0.0535	0.180	0.200	0.109
ν32	0.449	0.180	0.200	0.109
G12, GPa	4.30	2.90	4.50	1.54
G23, GPa	3.70	2.14	2.90	1.67
G31, GPa	4.30	2.14	3.10	1.67
X1T, MPa	1380	604	402	531
X1C, MPa	770	291	138	131
X2T, MPa	47	604	592	531
X2C, MPa	137	291	204	131
ХЗТ, МРа	47	58	52	50
SFC, MPa	850	850	1540	870
SFS, Mpa	250	300	172	160
S12, MPa	76	75	73	35
S23, MPa	38	58	49	27
S31, MPa	76	58	49	27
AM1	100.00	2.00	1.00	1.00
AM2	10.00	2.00	1.00	1.00
AM3	1.00	0.50	0.50	0.50
AM4	0.10	0.20	-0.20	0.20/-0.20
PHIC	10	10	10	10
SFFC	0.10	0.30	0.30	0.30
Crate1	0.00	0.03	0.03	0.03
Crate2	0.00	0.00	0.00	0.00
Crate3	0.00	0.03	0.03	0.03
Crate4	0.00	0.03	0.03	0.03
SDELM	1.2	1.2	1.2	1.2
OMGMX	0.999	0.999	0.998 (LVI) 0.997 (DoP	0.994
E_LIMT	4.5	4.5	5.0	4.0
EEXPN	4.5	4.5	5.0	4.0
ECRSH	0.001	0.001	0.700	0.500
AMODEL	1 (UD)	2 (PW)	2 (PW)	2 (PW)
SOURCE	Ref. [A]	Ref. [B]	Ref. [C]	Ref. [D]

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APPENDIX B: GENERAL DISCUSSION

DISCUSSION ON REFERENCE STRAIN RATE

The reference strain rate in MAT162 is set to 1. If the time unit in LS-DYNA computation is set to seconds (s), then this reference strain rate is $\dot{\bar{\epsilon}}_0 = 1s^{-1}$. This raises a question of how to calculate the rate parameters, C_{rate} , for different time units of LS-DYNA computations.

Consider a fictitious experimental set of strength data as a function of strain rates measured in s^{-1} . Table SR-1 shows this set of data. Fig. SR-1a shows the plot of this set of experimental data. The MAT162 rate equation is expressed as:

$$\frac{X_{RT}}{X_0} = 1 + C_{rate} \ln \left(\frac{\dot{\bar{\varepsilon}}}{\dot{\bar{\varepsilon}}_0} \right)$$
 (SR-1)

Table SR-1: A Fictitious Experimental Set of Strength Data and Dimensionless Data for Reference Strain Rate, $\dot{\bar{\epsilon}}_0 = 1.0 \times 10^{-6} \ s^{-1}$.

Strain Rate, s ⁻¹	Strength, MPa	$\dot{\bar{\varepsilon}}_0 = 1.0 \times 10^{-6} s^{-1}$	$X_0 = 600 MPa$
$\dot{ar{arepsilon}}$	X_{RT}	$\dot{\bar{\varepsilon}}/\dot{\bar{\varepsilon}}_0$	X_{RT}/X_0
1.00E-06	600	1.00E+00	1.0000
1.00E-05	630	1.00E+01	1.0500
1.00E-04	655	1.00E+02	1.0917
1.00E-03	680	1.00E+03	1.1333
1.00E-02	710	1.00E+04	1.1833
1.00E-01	740	1.00E+05	1.2333
1.00E+00	765	1.00E+06	1.2750
1.00E+01	790	1.00E+07	1.3167
1.00E+02	820	1.00E+08	1.3667
1.00E+03	850	1.00E+09	1.4167
1.00E+04	875	1.00E+10	1.4583

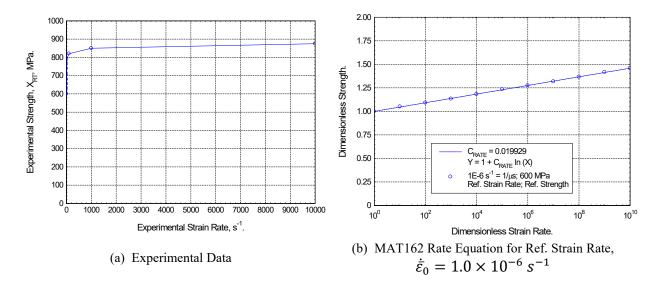


Figure SR-1: A Fictitious Experimental Set of Strength MAT162 Rate Equation for Reference Strain Rate, $\dot{\bar{\epsilon}}_0 = 1.0 \times 10^{-6} \ s^{-1}$.

We will use the Table SR-1 data to determine the rate parameter for different time units in LS-DYNA computations.

1. Reference Strain Rate $\dot{\bar{\epsilon}}_0=1.0\times 10^{-6}~s^{-1}=1/\mu s$ for LS-DYNA Time Unit of Micro-Second

Consider the LS-DYNA time unit be micro-second. Also consider the reference strain rate in micro-second time unit to be $\dot{\bar{\epsilon}}_0 = 1 \,\mu s^{-1} = 1.0 \times 10^{-6} \,s^{-1}$. From Table SR-1, the reference strength is $X_0 = 600 \, MPa$. We can then normalize the experimental data with the reference strain rate and reference strength and the dimensionless values $(\dot{\bar{\epsilon}}/\dot{\bar{\epsilon}}_0, X_{RT}/X_0)$ are also presented in Table SR-1. The dimensionless strength and strain rates can then be plotted and is presented in Fig. SR-1b. Eq. (SR-1) can then be used to fit the dimensionless experimental data in determining the rate parameter, C_{rate} , and for the fictitious experimental data presented in Table SR-1 is found to be, $[C_{rate}]_{\dot{\bar{\epsilon}}_0=1 \,\mu s^{-1}}=0.019929$.

2. Reference Strain Rate $\dot{\bar{\epsilon}}_0=1.0\times 10^{-3}~s^{-1}=1/ms$ for LS-DYNA Time Unit of Milli-Second

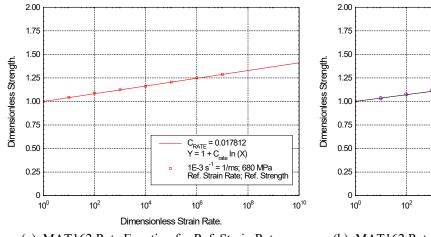
Consider the LS-DYNA time unit be milli-second. Also consider the reference strain rate in milli-second time unit to be $\dot{\bar{\epsilon}_0} = 1 \, ms^{-1} = 1.0 \times 10^{-3} \, s^{-1}$. The corresponding reference strength is $X_0 = 680 \, MPa$. Table SR-2 shows the dimensionless strain rate and stress data and is plotted in Fig. SR-2a. Note that strain rates $< 1.0 \times 10^{-3} \, s^{-1}$ are not considered. This data is curve fitted to determine the rate parameter and is found to be, $[C_{rate}]_{\dot{\bar{\epsilon}_0}=1 \, ms^{-1}} = 0.017812$.

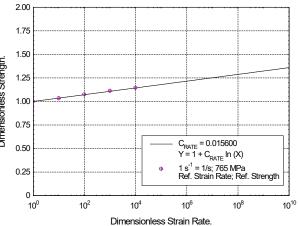
	Reference Strain Rate, $\varepsilon_0 = 1.0 \times 10^{-5} \text{ s}^{-1}$, and $\varepsilon_0 = 1.0 \text{ s}^{-1}$.								
Strain Rate, s ⁻¹	Strength, MPa	$\dot{\bar{\varepsilon}}_0 = 1.0 \times 10^{-3} \ s^{-1}$	$X_0 = 680 MPa$	$\dot{\bar{\varepsilon}}_0 = 1.0 \ s^{-1}$	$X_0 = 765 MPa$				
$\dot{ar{arepsilon}}$	X_{RT}	$\dot{ar{arepsilon}}/\dot{ar{arepsilon}}_0$	X_{RT}/X_0	$\dot{ar{arepsilon}}/\dot{ar{arepsilon}}_0$	X_{RT}/X_0				
1.00E-06	600	·	=	ı	=				
1.00E-05	630	·	=	ı	=				
1.00E-04	655	-	=	-	-				
1.00E-03	680	1.00E+00	1.0000	-	-				
1.00E-02	710	1.00E+01	1.0441	-	-				
1.00E-01	740	1.00E+02	1.0882	-	-				
1.00E+00	765	1.00E+03	1.1250	1.00E+00	1.0000				
1.00E+01	790	1.00E+04	1.1618	1.00E+01	1.0327				
1.00E+02	820	1.00E+05	1.2059	1.00E+02	1.0719				
1.00E+03	850	1.00E+06	1.2500	1.00E+03	1.1111				
1.00E+04	875	1.00E+07	1.2868	1.00E+04	1.1438				

Table SR-2: A Fictitious Experimental Set of Strength Data and Dimensionless Data for Reference Strain Rate, $\dot{\varepsilon}_0 = 1.0 \times 10^{-3} \ s^{-1}$, and $\dot{\varepsilon}_0 = 1.0 \ s^{-1}$.

3. Reference Strain Rate $\dot{\bar{\epsilon}}_0 = 1.0 \text{ s}^{-1} = 1/\text{s}$ for LS-DYNA Time Unit of Second

Consider the LS-DYNA time unit be second. Also consider the reference strain rate in second time unit to be $\dot{\bar{\epsilon}_0} = 1 \ s^{-1}$. The corresponding reference strength is $X_0 = 765 \ MPa$. Table SR-2 shows the dimensionless strain rate and stress data and is plotted in Fig. SR-2b. Note that strain rates < $1.0 \ s^{-1}$ are not considered. This data is curve fitted to determine the rate parameter and is found to be, $[C_{rate}]_{\dot{\bar{\epsilon}}_0 = 1 \ ms^{-1}} = 0.015600$.





(a) MAT162 Rate Equation for Ref. Strain Rate, $\dot{\bar{\varepsilon}}_0 = 1.0 \times 10^{-3} \ s^{-1}$

(b) MAT162 Rate Equation for Ref. Strain Rate, $\dot{\bar{\varepsilon}}_0 = 1.0 \ s^{-1}$

Figure SR-2: MAT162 Rate Equations for Reference Strain Rates, $\dot{\bar{\varepsilon}}_0 = 1.0 \times 10^{-3} \ s^{-1}$, and $\dot{\bar{\varepsilon}}_0 = 1.0 \ s^{-1}$.

DISCUSSION ON LAMIANTE ARCHITECTURE AND PREDEFINED DELAMINATION PLANES

A composite laminate may contain several numbers of laminas or plies stacked through-the-thickness of the laminate. If the individual laminas are very thin, it is suggested to combine several laminas into one sub-laminate. Figure LA-1 shows the finite element model of such a sub-laminate with three (3) through-thickness elements.

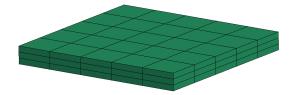


Figure LA-1: Finite Element Model of a Sub-Laminate with Three (3) Through-Thickness Elements.

Once a sub-laminate model is created, one should assign a Part ID to the sub-laminate and associate the PID with a Material ID with a pre-defined material angle (BETA), e.g.; PID=1, MID=100, BETA=0 in Fig. LA-2. Several sub-laminates can be stacked through-the-thickness to build a laminate. Figure LA-2 shows a composite laminate with four (4) sub-laminates stacked through-thickness with the stacking sequence [0/90/0/90] and each sub-laminates are assigned with different Part IDs, i.e., PID=1, 2, 3, 4. However, since the stacking sequence is taken as [0/90/0/90], only two Material IDs (i.e., MID=100, 200) are sufficient. Note that all duplicate nodes between the PIDs 1 to 4 needs to be merged and the unreferenced nodes be deleted.

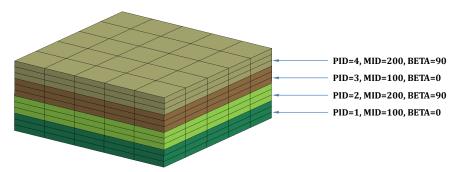


Figure LA-2: Node Merged Finite Element Model of a Laminate consisting of Four (4) Sub-Laminates with the Stacking Sequence [0/90/0/90]. Three Delamination Planes are thus Predefined between PIDs 1&2, 2&3, and 3&4.

According to MAT162 formulations, three pre-defined delamination planes will be automatically defined at the interface between four parts with different material angles (**BETA**). Once delamination between two parts with different material angles is predicted, shear properties of

the elements adjacent to the delamination interface will be degraded to mimic delamination without creating physical surfaces between sub-laminates or parts. The advantage of MAT162 delamination criterion is that it is simple, however, the disadvantage is that the predicted delamination is assigned a thickness equal to one element near the delamination interface and is not physical in nature. This is why, three elements through-the-thickness of a sub-laminate or part is proposed so that the pseudo-delamination is limited to one-third of the thickness of a sub-laminate or part.

Figures LA-3 to 5 show three additional laminate stacking sequences defined with four sub-laminates.

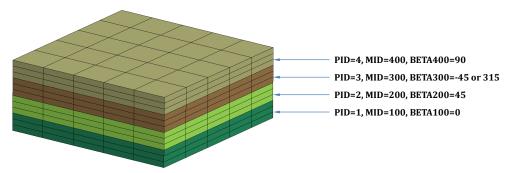


Figure LA-3: Node Merged Finite Element Model of a Laminate consisting of Four (4) Sub-Laminates with the Stacking Sequence [0/45/-45/90]. Three Delamination Planes are thus Predefined between PIDs 1&2, 2&3, and 3&4.

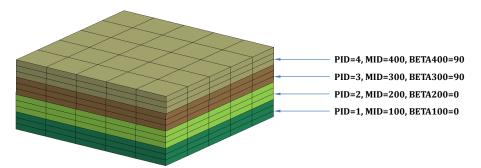


Figure LA-4: Node Merged Finite Element Model of a Laminate consisting of Four (4) Sub-Laminates with the Stacking Sequence [0/0/90/90]. One Delamination Plane is thus Predefined between PIDs 2&3.

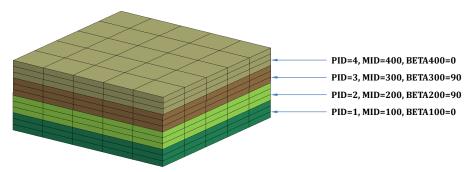


Figure LA-5: Node Merged Finite Element Model of a Laminate consisting of Four (4) Sub-Laminates with the Stacking Sequence [0/90/90/0]. Two Delamination Planes are thus Predefined between PIDs 1&2, and 3&4.

DISCUSSION ON CONTROL ACCURACY

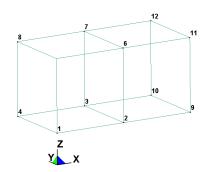


Figure CA-1: Definition of Two Single Elements in the Discussion of Control Accuracy.

In Fig. CA-1, if we choose **AOPT=0** for a composite element and define the element (**EID=1**) with the following node sequence, then the vector **V12** connecting nodes 1 & 2 defines the material direction 1 or A. The vector **V14** connecting nodes 1 & 4 together with the vector **V12** defines the material plane 1-2 or A-B, and the vector cross product **V12** × **V14** defines the through-thickness material direction 3 or C.

*EL	EMENT_S									
\$#	eid	+ pid	n1	n2	n3	n4	n5	n6	n7	n8
\$	1	100	1		3	4	5	6	7	8

Following the same procedure, one can define the second element (EID=2) as follows to be consistent with AOPT=0.

\$	+	+	+	+	+	+	+	+	+	+
*EL	EMENT_S	OLID								
•		+								
•		pid							n7	
Ş	2	+ 100	_		_	2		12	7	
\$	_ +	+	-		-	_		+		

However, if the second element (**EID=2**) is defined as follows, then it is not consistent with **AOPT=0**.

\$++++++										
			+	+	+	+	+	+	+	+
\$#	eid		n1	n2	n3	n4	n5	n6	n7	n8
Ş	1	100	1	_	3	4	5	6	7	8
	2	100	2	9	10	3	6	11	12	7
\$	+	+	+	+	+	+	+	+	+	+

One can try to fix this problem by choosing **AOPT=2** by defining two vectors **A**(1,0,0) & **B**(0,1,0) to represent material direction 1 or A and the plane 1-2 or A-B, but this will not correct the problem, instead produce a wrong material response.

Figure CA-2a shows the wrong stress-time plot using **AOPT=2** with the wrong element definition for **EID=2**. In order to solve this problem, the LS-DYNA control card ***CONTROL_ACCURACY** can be used which allows **INVARIENT NODE NUMBERING** if the parameter **INN=3** is chosen for solid elements. Figure CA-2b shows the correct stress-time response with **INN=3** in the control accuracy card.

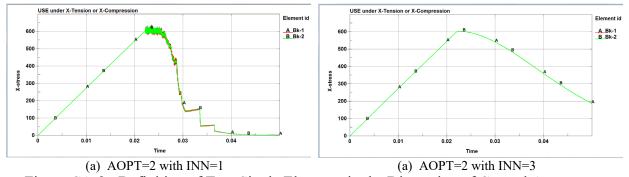


Figure CA-2: Definition of Two Single Elements in the Discussion of Control Accuracy.

		_		_	ACCURA		_		PARAMETI	ERS
*PARAN	METER	_EXPRE	SSION		•				+	
\$ PRMI	R1 :	EXPRES	SION1		+					
i inn		3								
\$	+		+		+				-+	+
\$+*CONTROL_ACCURACY \$+										
\$#	osu 0		inn &INN	pidos	•	+	+		-+	+
ş	+		+		+	+	+		-+	+

AXIS OPTION AOPT AND SWITCHING MATERIAL AXES

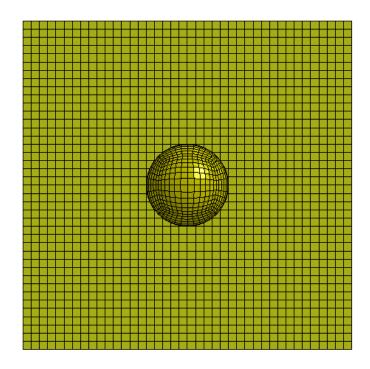
Axis options, AOPT=0, 2, 4 has been tested and are found to be working with MAT162 along with the material axes switch options MACF without any problems. At present, we are looking at the applicability of axis option AOPT=1, and will report when our test is complete.

APPENDIX C: MAT162 EXAMPLES

EXAMPLE 1: SPHERE IMPACT ON COMPOSITES An Example of MAT162 Progressive Composite Damage Model

A SIMPLE FEA MODEL OF A FOUR LAYER COMPOSITE LAMINATE WITH ONE DELAMINATION PLANE

The FEA model has two parts, each part defining two layers/laminas of a four layers composite laminate. The two bottom layers/laminas with PART ID = 101 is assigned a material angle of zero degree by defining the parameter BETA101 = 0.00 and the same for the top part with PART ID = 102 with the parameter BETA102 = 90.00. A delamination plane is thus defined between two different PART IDs each having a different material angle defined by BETA. For example, if the parameter BETA102 is changed to zero degree, there will be no delamination plane defined in the model.



The input files used in this example are summarized below.

FILE: 001-Sphere-Impact-on-Composite-Plate.key

```
$ Copyright 2015 (C) University of Delaware Center for Composite Materials
   $ Unit System: mm-tonne-s (milimeter-tonne-second-Newton-MegaPascal)
$ Date: April 15, 2015
$ Dr. Bazle Z. (Gama) Haque
$ Senior Scientist of Center for Composite Materials (UD-CCM)
$ Assistant Professor of Mechanical Engineering
$ University of Delaware
$ Newark, DE 19716, USA
$ Tel: (302) 690-4741; E-mail: gama@udel.edu
$ Visualizing Damage Modes in LS-PrePost
$ Page-1 Tab-Range: User Min=0 Max=1
$ Toggle Switch: Lcon-ON
$ Page-1 Tab-Fcomp: Misc history var#7 (Fiber Tension-Shear Along A)
$ Page-1 Tab-Fcomp: Misc history var#8 (Fiber Tension-Shear Along B)
$ Page-1 Tab-Fcomp: Misc history var#9
                        (Fiber Crush)
$ Page-1 Tab-Fcomp: Misc history var#10 (In-Plane Matrix Crack)
$ Page-1 Tab-Fcomp: Misc history var#11 (Transverse Matrix Crack)
$ Page-1 Tab-Fcomp: Misc history var#12 (Delamination between 0 & 90 pliies)
$ Delamination between 0 & 90 pliies for this specific problem
\# Change the NCPU below to match your desired number of CPUs.
*KEYWORD MEMORY=10M NCPU=16
*PARAMETER EXPRESSION
$-----+
R ENDTIME 600.00*us
R NOD3 30.0
R DTBIN 0.01*us
R DTD3 ENDTIME/NOD3
R TSSFAC 0.950
$-----+
T BSORT 1
I DEPTH
R PARMAX 1.0025
I SBOPT 5
R SLSFAC 0.010
R SLDTHK 0.001
I NEIPH 36
           -+----+
R mps 1000.00
   +0.001*mps
+0.001*mps
R VY
     -100.00*mps
R mm 1.000
R dx 0.01*mm
    0.01*mm
12.10*mm
R dy
$-----+
$ *** MAT162 PROPERTIES & PARAMETERS ***
R gmpcc 1.0E-9
     1.000
R MPa
R GPa
     1.0E+3
T MTDP 700
R ROP 7.85*gmpcc
R EP 210.00*GPa
R PRP 0.290
$-----+
I MID101 101
I MID102 102
R BETA101 0.00
```

\$+	+
R NU21 0.110 R NU31 0.180 R NU32 0.180 \$+++++	+
R G12	+
R X1T 604.0*MPa R X1C 291.0*MPa R X2T 604.0*MPa R X2C 291.0*MPa R X3T 58.0*MPa \$+ R SFC 850.0*MPa	+
R SFC 850.0*MPa	+
	+
\$+++++++	
R SFFC 0.300 R PHIC 10.00 R SDELAM 1.200 \$+	
R OMGMX 0.999 R ECRSH 0.001 R ELIMIT 4.000 R EEXPN 4.000 \$+	
R AM1FD1 2.00 R AM2FD2 2.00 R AM3FCPS 0.50 R AM4MCDE 0.20	
\$+	
\$++++++	+
Sphere Projectile S+	
\$# mid ro e pr da db not used &MIDP &ROP &EP &PRP	
\$++++++	
PW S-2 Glass/SC-15 0 Layer with beta = 0	
\$# mid ro ea eb ec prba prca &MID101 &DENSITY &E11 &E22 &E33 &NU21 &NU31	prcb &NU32
\$# gab gbc gca aopt macf &G12 &G23 &G31 2.0 0	411002
\$# xp yp zp a1 a2 a3 0.000 0.000 0.000 1.000 0.000 0.000	
\$# v1 v2 v3 d1 d2 d3 beta	
\$# sat sac sbt sbc sct sfc sfs	sab
\$\frac{\pi \text{8X1T}}{\pi \text{8X1C}} \ \frac{\pi \text{X2T}}{\pi \text{8X2C}} \ \frac{\pi \text{X3T}}{\pi \text{8SFC}} \ \frac{\pi \text{SFS}}{\pi \text{8SFS}} \ \frac{\pi \text{8SFC}}{\pi \text{8SFS}} \ \frac{\pi \text{8X2T}}{\pi \text{8SFC}} \ \frac{\pi \text{8X3T}}{\pi \text{8SFC}} \ \frac{\pi \text{8X1T}}{\pi \text{8SFC}} \ \frac{\pi \text{8X1T}}{\pi \text{8X2T}} \ \frac{\pi \text{8SFC}}{\pi \text{8X2T}} \ \frac{\pi \text{8X2T}}{\pi \text{8X2T}} \ \frac{\pi \text{8SFC}}{\pi \text{8X2T}} \ \frac{\pi \text{8SFC}}{\pi \text{8X2T}} \ \frac{\pi \text{8X2T}}{\pi \text	&S12

\$# \$#	omgmx &OMGMX am2/fdb		eexpn &EEXPN am4/mc/de	cr1/sig &CR1SIG cr2/ea/eb	am1/fda &AM1FD1 cr3/g	cr4/ec		
	&AM2FD2		&AM4MCDE	&CR2E1E2	&CR3G	&CR4E3		+
*MA	T COMPOS	SITE DMG MS	SC TITLE	+				
PW	S-2 Glas	ss/SC-15 0	Layer with	beta = 90				+
\$ \$#	 mid	 ro	 ea		 ec	+- prba	prca	 prcb
	&MID102	&DENSITY	&E11	&E22	&E33	&NU21	&NU31	&NU32
\$#	gab &G12	gbc &G23	gca &G31	aopt 2.0	macf 0			
\$#	qx 0.000	ур 0.000	zp 0.000	a1 1.000	a2 0.000	a3 0.000		
\$#	v1	v2	v3	d1	d2	d3	beta	
\$#	0.000 sat	0.000 sac	0.000 sbt	0.000 sbc	1.000 sct	0.000 sfc	&BETA102 sfs	sab
ΥII	&X1T	&X1C	&X2T	&X2C	&X3T	&SFC	&SFS	&S12
\$#	sbc	sca	sffc	amodel	phic	e_limt	s_delm	
\$#	&S23 omgmx	&S31 ecrsh	&SFFC eexpn	2.0 cr1/sig	&PHIC am1/fda	&ELIMIT	&SDELAM	
7 11	&OMGMX	&ECRSH	&EEXPN	&CR1SIG	&AM1FD1			
	am2/fdb &AM2FD2		am4/mc/de &AM4MCDE	cr2/ea/eb &CR2E1E2	cr3/g &CR3G	cr4/ec &CR4E3		
				&CRZEIEZ			+-	+
			ER EXPRESSI	ONS ***				
*IN	CLUDE							
			+-	+	+-	+-	+-	+
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\$	_	_	_	+	+-	+-	+-	+
		OURGLASS	+-				+	+
\$#	ihq 7	qh 0.100000						
	NTROL_OU		+-					+
\$#	npopt 0	neecho 0	nrefup 0	iaccop 0	opifs 0.000	ipnint 0	ikedit 100	iflush 100
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	0	1	2	50				
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*CO	NTROL_T	IMESTEP					+-	+
\$#	dtinit 0.000	tssfac &TSSFAC	isdo					
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*DA	TABASE (GLSTAT	· +-					
\$#	dt &DTBIN	binary 3		ioopt	dthff	binhf		
*DA	TABASE_N	MATSUM	+-					
\$#	dt &DTBIN	binary 3		ioopt	dthff	binhf		
*DA	TABASE_F	RCFORC						
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*DA	TABASE_E	BINARY_D3PI						
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	ioopt 0							
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\$#	cmpflg	ieverp	beamip				n3thdt	
	nintsld 0	pkp_sen 0	sclp 1.000000	unused	msscl	therm		
*CO	NTACT_FO	DRCE_TRANSI	OUCER_PENAL	TY_ID				
\$#	cid 700	title Sphere Imp				+-	+-	+
\$ \$#	ssid	msid	sstyp					
\$#	700 fs	fd		VC	vdc	penchk		dt
\$#	0.000 sfs	sfm	sst	mst	0.000 sfst	sfmt	fsf	
\$		+-	0.000					
	T_PART_I		+-	+-	+-	+-	+-	+
\$#	sid 123		da2	da3	da4			
\$#	pid1 101	102	pid3 700	-	-	-	-	-
*SE	T_PART_I	LIST	+-					
\$ \$#	sid	da1				+-	+-	+
\$#	700 pid1 700		pid3	pid4	pid5	pid6	pid7	pid8

*PA	RT							
					+-			+
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\$ *PA				+-				+
\$	+-	+-		+-	+-	+-	+-	+
	-	ayer 2 BOT					, ,	
\$# \$	102		mid 102 +	eosid +-	hgid	grav	adpopt 	tmid +
* PA	RT							
	ere Proj		'	'	,			·
\$#	pid 700	100	mid 700	eosid	_	-		tmid
*SE	CTION_SC	DLID						+
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ςπ	&SLSFAC	0.000	1	0	0	0	1	1
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\$#		nserod			rwksf	icov		ithoff
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\$#	fs		dc	VC	vdc	penchk		
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\$#	isym 0		iadj 1	J. 300	3.00000	3.00000	3.00000	2.230000
\$#	soft			maxpar	sbopt	depth	bsort	frcfrq
	2	&SLSFAC	0	&PARMAX	sbopt &SBOPT	&DEPTH		1
\$#	penmax	thkopt	shlthk	snlog	isym O	i2d3d 1		sldstf 0.000
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\$	+-	+-	+	+-	+-	+-	+-	+

FILE: 003-101-102-80mmx80mmx2mm-Composite-Plate-FEM.key

This file contains the *NODE & *ELEMENT_SOLID cards for the composite parts 101 and 102.

FILE: 004-700-Sphere-FEM.key

This file contains the *NODE & *ELEMENT SOLID cards for the Steel sphere part 700.

\$								+
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\$								+
*BC	UNDARY SI	PC SET						
\$#	\overline{nsid}	- cid	dofx	dofy	dofz	dofrx	dofry	dofrz
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	1000001	1000002	1000003	1000004	1000005	1000006	1000007	1000008
	1000009	1000010	1000011	1000012	1000013	1000014	1000015	1000016
	1008390	1008391	1008392	1008393	1008394	1008395	1008396	1008397
	1008398		1008400	1008401		1008403		
								+
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\$						+-		+

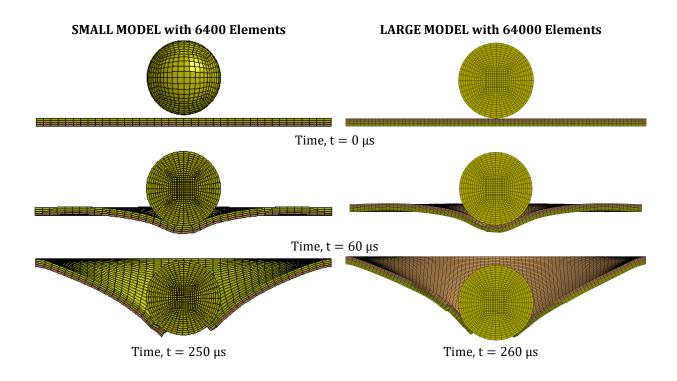
RUNNING A LARGER MODEL

Above files runs a smaller model with 40x40x4=6400 elements for the composite plate. There are two additional files that will run a larger model with 80x80x10=64000 elements for the composite plate. Name of these two files are as follows:

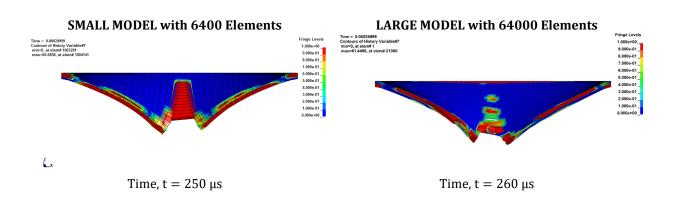
003-101-102-80mmx80mmx2mm-Composite-Plate-FINE-FEM.key 005-1000-FINE-SPC.key

RESULTS AND DISCUSSION

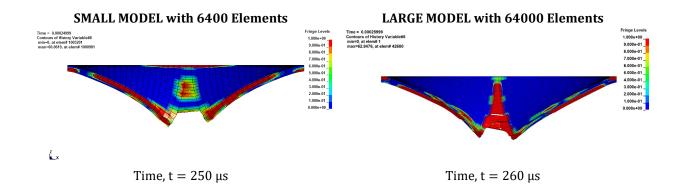
Deformation as a Function of Time



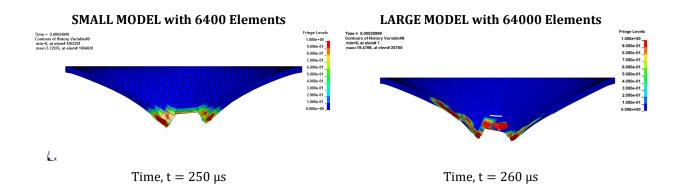
Contour of History Variable #7, Fiber Tension-Shear Damage Mode along A



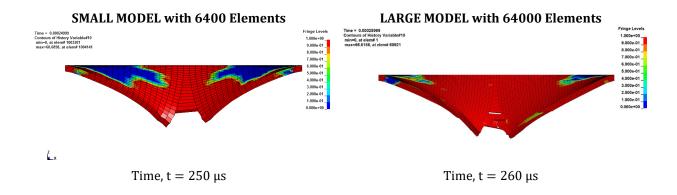
Contour of History Variable #8, Fiber Tension-Shear Damage Mode along B



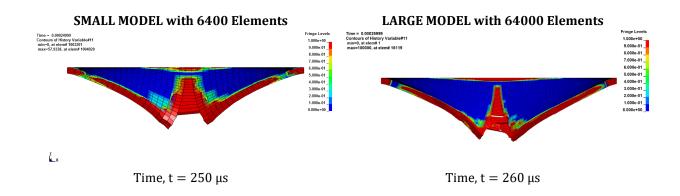
Contour of History Variable #9, Fiber Crush Damage Mode



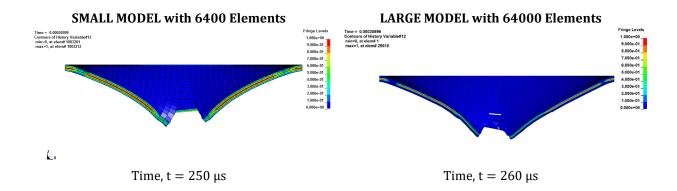
Contour of History Variable # 10, In-Plane Matrix Damage Mode



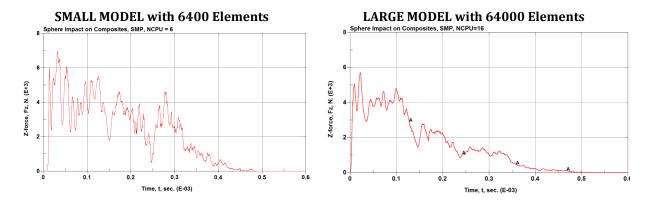
Contour of History Variable # 11, Transverse Matrix Damage Mode



Contour of History Variable # 12, Delamination Damage Mode



Time History of Impact-Contact Force



This example problem can be downloaded from UD-CCM MAT162 website: http://www.ccm.udel.edu/software/mat162/

EXAMPLE 1: Sphere-Impact-on-Composites-SOLID-MAT162.zip

```
This zip folder contains the following KEWORD files. Run the first file. 001-Sphere-Impact-on-Composite-Plate.key 002-contact-SMP.key 003-101-102-80mmx80mmx2mm-Composite-Plate-FEM.key 003-101-102-80mmx80mmx2mm-Composite-Plate-FINE-FEM.key 004-700-Sphere-FEM.key 005-1000-FINE-SPC.key 005-1000-SPC.key
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EXAMPLE 2: SPHERE IMPACT ON COMPOSITE SHELLS & SOLIDS Comparing Composite Damage Modeling with MAT-162 and MAT-054

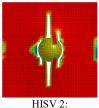
MAT162 is a rate dependent progressive composite damage model for solid elements. In composite shell elements, transverse shear damage cannot be defined and this is why a composite solid element defining transverse shear damage is essential in modeling problems where transverse shear damage is important. Shell elements are efficient for thin-section composites, whereas thick-section composites can be modeled using multiple through-thickness elements. MAT162 for solid elements have unique quadratic progressive damage modes, which no other material model offers. These unique damage modes, such as, (i) Tension-Shear, (ii) Axial Compression, (iii) Inter laminar Shear (ILS) and Delamination, (iv) Fiber Shear (FS) or Punch Shear (PS), and (v) Fiber Crush provides the theoretical framework for modeling most known composite damages.

Benefits of MAT162 progressive damage can be elucidated with simple examples. Consider the transverse impact of a 20mm sphere with an impact velocity of 50 m/s on an 80mm x 80mm x 2mm composite plate modeled with perfectly clamped boundary condition. Shell elements with MAT054 and solid elements with MAT162 damage predictions shows that MAT054 can only model in-plane tension and compression damage, while MAT162 can predict tension-shear along direction 1 & 2, in-plane shear, transverse shear and associated delamination.

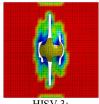
Damage Modes: Shell Elements – MAT054



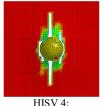
HISV 1: Tensile Fiber Mode



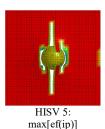
Compres. Fiber Mode



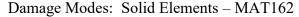
Tensile Matrix Mode

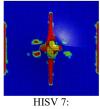


Compres. Matrix Mode

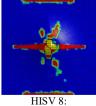


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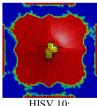




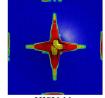
HISV 7: Tension-Shear Mode 1



Tension-Shear Mode 2



In-Plane Matrix Mode



HISV 11: Transverse Matrix Mode



HISV 12: Delamination Index

In modeling out-of-plane deformation of any kind, the primary damage mode is tension-shear where MAT162 defines a quadratic damage function with in-plane tension and out-of-plane shear properties for the initiation of tension-shear damage.

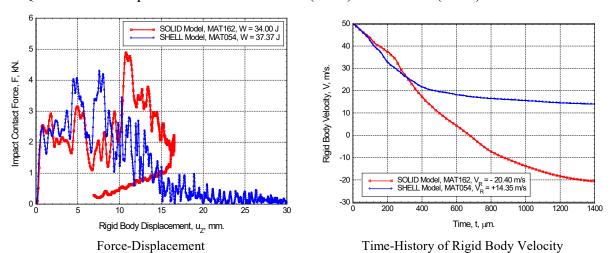
$$\mathbf{f_7} = \left\{ \frac{\mathbf{E_1} \langle \boldsymbol{\epsilon_1} \rangle}{\mathbf{X_1^T}} \right\}^2 + \left\{ \frac{\mathbf{G_{31}} \boldsymbol{\epsilon_{31}}}{\mathbf{S_1^{FS}}} \right\}^2$$

The MAT054 equivalent damage function uses in-plane tension and in-plane shear properties instead, and in general fails to account for transverse shear deformation.

$$e_{f} = \left\{ \frac{E_{1}\epsilon_{1}}{X_{1}^{T}} \right\}^{2} + \beta \left\{ \frac{G_{12}\epsilon_{12}}{S_{12}} \right\}$$

The inability of MAT54 in modeling transverse damage modes greatly affects the impact-contact force and associated work done, predicts premature perforation while MAT162 predicts partial perforation and complete rebound of the sphere in case of initial impact velocity of 50 m/s.

Quantitative Comparison between MAT054 (Shell) & MAT162 (Solid) Material Models



A local MAT162 solid model in the anticipated damage region combined with global shell elements can be used to efficiently and accurately model large-scale impact application such as, crash analysis of automotive composite components and aerospace structures under impact conditions.

This example problem can be downloaded from UD-CCM MAT162 website: http://www.ccm.udel.edu/software/mat162/

EXAMPLE 1: Sphere-Impact-on-Composites-SOLID-MAT162.zip

```
This zip folder contains the following KEWORD files. Run the first file. 001-Sphere-Impact-on-Composite-Plate.key 002-contact-SMP.key 003-101-102-80mmx80mmx2mm-Composite-Plate-FEM.key 003-101-102-80mmx80mmx2mm-Composite-Plate-FINE-FEM.key 004-700-Sphere-FEM.key 005-1000-FINE-SPC.key 005-1000-SPC.key
```

EXAMPLE 2: Sphere-Impact-on-Composites-SHELL-MAT54.zip

This zip folder contains the following KEWORD files. Run the first file.

- 001-Sphere-Impact-on-Composite-Shell.key
- 002-control-cards.key
- 003-Shell-Plate-80mm-x-80-mm-FEM.key
- 004-700-Sphere-FEM.key
- 005-parts-partsets-sections-materials-FEM.key
- 006-100-Node-SET.key
- 006-nodeset-elementset-segmentset-FEM.key
- 007-contact-single-surface.key
- 008-boundary-conditions.key
- 009-initial-conditions.key
- 010-loading-conditions.key
- 011-output-database.key